

# FUNDAMENTALS OF SPUTTER DEPOSITION

Ivan Petrov

Materials Research Laboratory, University of Illinois at Urbana-Champaign, USA  
Department of Physics, Chemistry, and Biology, Linköping University, Sweden

petrov@illinois.edu

RWTH Aachen University  
February 10, 2025

1

## REVIEW ARTICLE

---



### Review Article: Tracing the recorded history of thin-film sputter deposition: From the 1800s to 2017

J. E. Greene<sup>a)</sup>

*D. B. Willett Professor of Materials Science and Physics, University of Illinois, Urbana, Illinois, 61801;  
Tage Erlander Professor of Physics, Linköping University, Linköping, Sweden, 58183, Sweden; and University  
Professor of Materials Science, National Taiwan University Science and Technology, Taipei City, 106, Taiwan*

Citation: *Journal of Vacuum Science & Technology A: Vacuum, Surfaces, and Films* **35**, 05C204 (2017); doi:  
10.1116/1.4998940

View online: <https://doi.org/10.1116/1.4998940>

**526 references; open access**

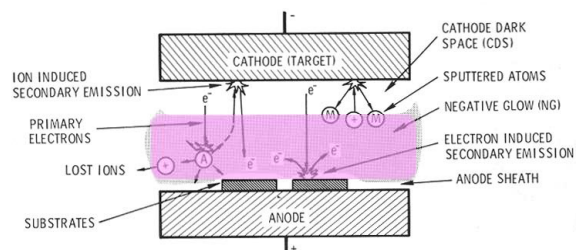
### D. Conclusions

It seems perfectly safe to predict that sputter-deposition of thin films will remain a vibrant and active field of human endeavor incorporating fundamental scientific research, process development, deposition-system design, and new-product manufacturing for periods far into the future.

2

2

## Sputter deposition



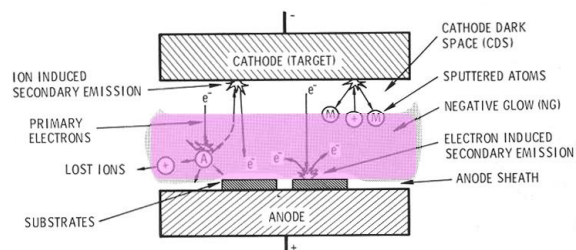
### Basics:

- A voltage is applied across a rarified gas in a vacuum.
- Breakdown of the gas forms a glow discharge plasma.
- Positive ions from the plasma strike the negative electrode (cathode/target).
- Energy from the ions is transferred to target atoms.
- A few of these may escape from the target surface (they are sputtered).
- Sputtered atoms undergo collisions in the gas phase
- Some of the sputtered atoms reach the substrate
- The sputtered atoms condense on the substrate forming a film.

3

3

## Sputter deposition



### Important processes:

- Ion-solid interactions on the target.
- Plasma generation and discharge maintenance
- Collisions in the gas phase – ionization, scattering.
- Nucleation and film growth on the substrate.
- Use of ion solid interactions to modify film growth

4

4

## Part 1. Fundamentals of Sputter Deposition

### 1.1 Elements of kinetic theory of gases

- Gas laws
- mean free path
- gas impingement rate

### 1.2 Elements of plasma physics

- Plasma probes
- Sheath width
- Penning ionization
- Electron energy distribution functions

### 1.3 Glow discharge maintenance

- Secondary ion-electron emission
- Electron ionization cross-sections

### 1.4 Sputtering yield

- Linear cascade model
- Correction for threshold effects
- Sputtering efficiency
- Energy of sputtered atoms
- Other energetic particles: backscattered ions and negative oxygen ions

### 1.5 Transport in the gas phase

- Thermalization
- Deposition rate calculation

### 1.6 Sputtering systems

- Magnetron sputtering
- Reactive sputtering

### 1.7 HIPIMS

- Source of metal ions
- Time separation between gas and metal ions
- High energy ions
- Lower deposition rates
- Bipolar HIPIMS
- Control of doubly charged ions

5

5

## Elements of Kinetic Theory of Gases

6

6

For gas phase transport we need some basic formulas  
from the **kinetic theory of gases**

Mean free path  $\lambda = \frac{1}{\sqrt{2} n \sigma}$

Gas density  $n = \frac{p}{kT}$

Average velocity  $\bar{v} = \sqrt{\frac{8kT}{\pi m}}$

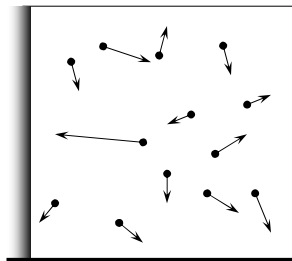
Impinging gas flux on the walls  $J_g = \frac{\bar{nv}}{4} = \frac{p}{\sqrt{2\pi mkT}}$

Modifications of these expressions are used in plasma physics

7

## Kinetic theory of gases

provides microscopic picture of gas laws



### Assumptions:

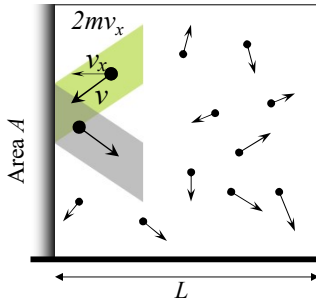
- A gas consists of a large number of molecules;
- separations are large compared to molecule size
- Molecules move randomly with a distribution in velocities which remains constant
- Molecules obey Newton laws of motion.
- Elastic collisions between the molecules and between molecules and the walls; no forces otherwise.

8

8



## Pressure of gas molecules



$$n = \frac{N}{V}$$

$$p = F/A \text{ units [N/m}^2\text{] or [Pa]}$$

$$\text{Momentum change in one collision} - \Delta p = 2 m v_x$$

$$\text{Time between collisions} - \Delta t = 2L/v_x$$

$$\text{Force by one molecule} - \Delta p/\Delta t = m v_x^2/L$$

$$\text{Force by N molecules} - N m v_x^2/L$$

$$\text{Pressure} - p = N m v_x^2/L A = n m v_x^2$$

However, the average kinetic energy in x

$$E = \frac{1}{2} m v_x^2 = \frac{1}{2} kT$$

$$p = n k T$$

**Ideal gas laws:**

Boyle:  $p \sim 1/V$

Charles:  $V \sim T$

Gay-Lussac:  $p \sim T$

$N$  - number of molecules

$V$  - volume ( $= L \times A$ )

$p$  - pressure

$T$  - absolute temperature

$m, v$  - mass and velocity of molecules

$k$  - Boltzmann constant =  $1.38 \times 10^{-23}$  [J/K]

9

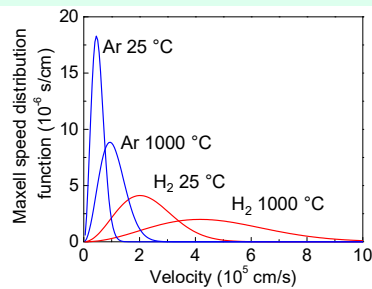
9

## Velocity and energy distribution of gas molecules

### Maxwell-Boltzmann distribution

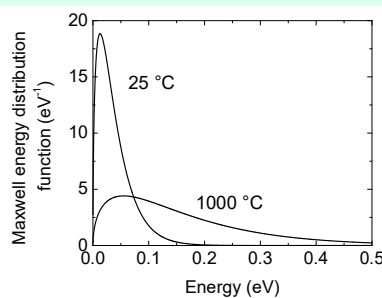
#### Velocity

$$f(v) = 4\pi \left(\frac{m}{2\pi kT}\right)^{\frac{3}{2}} \exp\left(-\frac{m}{2kT}v^2\right) v^2$$



#### Energy

$$f(e) = 2\pi \left(\frac{1}{\pi kT}\right)^{\frac{3}{2}} \exp\left(-\frac{e}{kT}\right) \sqrt{e}$$



**average**

$$v_{av} = \sqrt{\frac{8}{\pi}} \sqrt{\frac{kT}{m}}$$

**root-mean-square**

$$v_{rms} = \sqrt{3} \sqrt{\frac{kT}{m}}$$

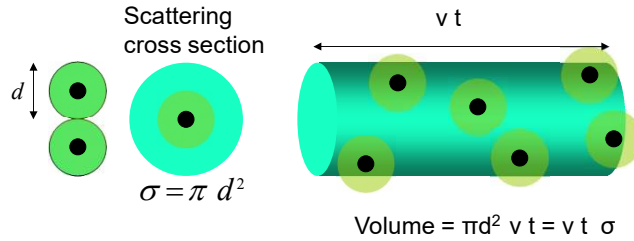
**most probable**

$$v_p = \sqrt{2} \sqrt{\frac{kT}{m}}$$

10

10

## Collisions and mean free path



$$\text{Mean free path} \gg \frac{\text{travelling distance}}{\text{number of collisions}} = \frac{v t}{n \pi d^2 v t} = \frac{1}{n \sigma}$$

More accurate expression:  
 (with relative motion of all gas molecules)

$$l = \frac{1}{\sqrt{2} n \sigma}$$

11

11

## Some useful formulas from kinetic theory of gases

Mean free path  $\lambda = \frac{1}{\sqrt{2} n \sigma}$

Gas density  $n = \frac{p}{kT}$

Average velocity  $\bar{v} = \sqrt{\frac{8kT}{\pi m}}$

Impinging gas flux on the walls  $J_g = \frac{\bar{v}}{4} = \frac{p}{\sqrt{2\pi} m k T}$

Deposition flux  
 growth rate R [cm/s]  $J_{dep} = R n_{dep}$   
 films density  $n_{dep}$  [at/cm<sup>3</sup>]

Note: from the bottom two expressions one can calculate the expected contamination of the films due to residual gases from the background pressure and the deposition rate  $J_g/J_{dep}$

12

12

## Examples

	$\lambda$		$J$ ( $\text{cm}^{-2} \text{s}^{-1}$ )		$t_{\text{ML}}$
	25°C	1000°C	25°C	1000°C	25°C
1 atm ( $10^5$ Pa)	100 nm	400 nm	$2.4 \times 10^{23}$	$1.2 \times 10^{23}$	4 ns
1 Torr (133 Pa)	70 $\mu\text{m}$	300 $\mu\text{m}$	$3.2 \times 10^{20}$	$1.6 \times 10^{20}$	3.0 $\mu\text{s}$
1 mTorr (0.133)	7 cm	30 cm	$3.2 \times 10^{17}$	$1.6 \times 10^{17}$	3.0 ms
$10^{-6}$ Torr	70 m	300 m	$3.2 \times 10^{14}$	$1.6 \times 10^{14}$	3.0 s
$10^{-11}$ Torr	7,000 km	30,000 km	$3.2 \times 10^9$	$1.6 \times 10^9$	84 hours

$\lambda$ : mean free path

$J$ : flux of the atoms on sample/walls

$t_{\text{ML}}$ : time to form a monolayer (ML) at sticking probability of 1

diameter of a gas atom  $\sim 3 \times 10^{-8}$  cm; area of a gas atom  $\sim 10^{-15}$   $\text{cm}^2$   
 one monolayer  $\sim 10^{15}$  atoms  $\text{cm}^{-2}$

For comparison: Ti growth with 1  $\mu\text{m}/\text{h}$       $J_{\text{Ti}} = 1.6 \times 10^{15}$  ( $\text{cm}^{-2} \text{s}^{-1}$ )

13

13

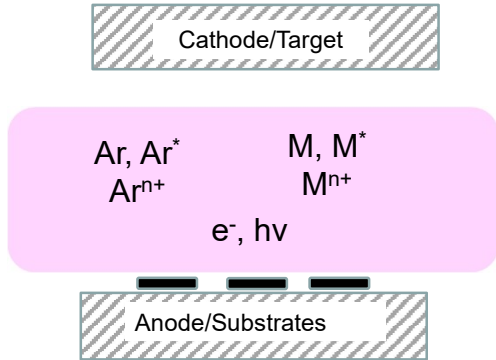
## Elements of Plasma Physics

14

14

# Sputter deposition plasmas

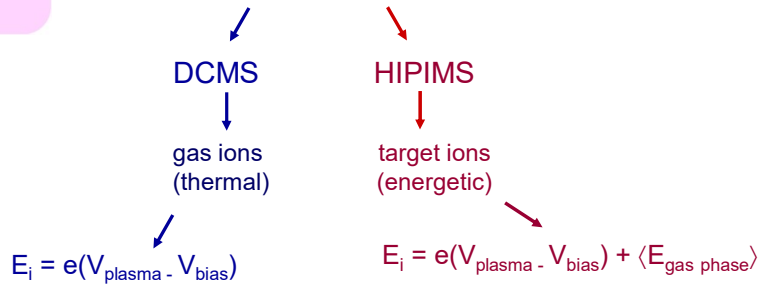
Plasma



Many plasma particles:

- neutral atoms, gas and target
- electrons
- ions, gas and target
- excited atoms, gas and target
- plasma radiation

Which are most important plasma species for film growth?

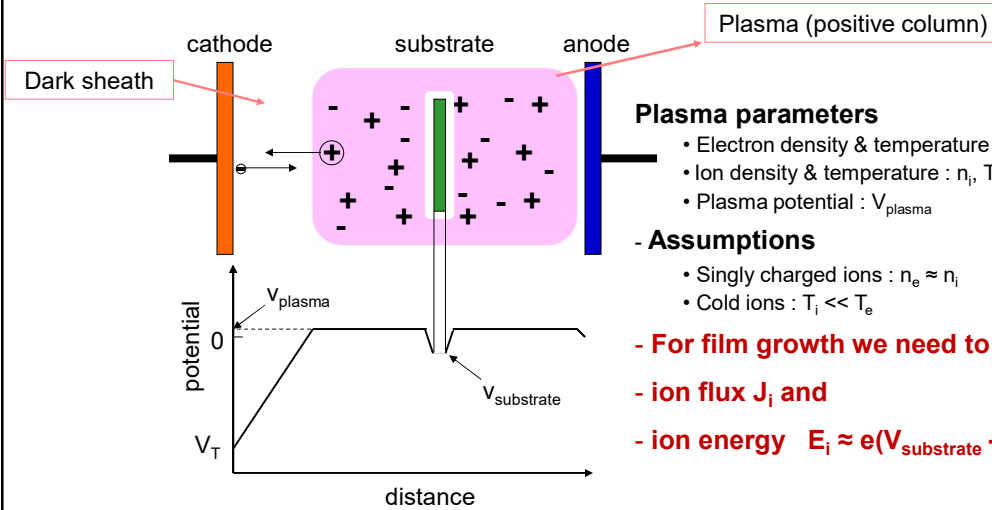


15

15

# Plasma characterization by electrostatic probes

Plasma



## Plasma parameters

- Electron density & temperature :  $n_e, T_e$
- Ion density & temperature :  $n_i, T_i$
- Plasma potential :  $V_{plasma}$

## - Assumptions

- Singly charged ions :  $n_e \approx n_i$
- Cold ions :  $T_i \ll T_e$

- For film growth we need to know:

- ion flux  $J_i$  and

- ion energy  $E_i \approx e(V_{substrate} - V_{plasma})$

16

16

# Why is the dark sheath dark?

Because electron leave the cathode with energy well above the maximum of the excitation cross section

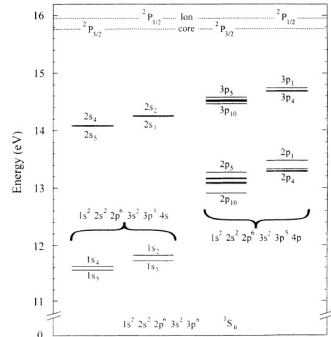


FIG. 1. Energy-level diagram of the argon atom. The Paschen notation ( $1s_2, 1s_3, 1s_4, 1s_5$  and  $2p_1, 2p_2, \dots, 2p_{10}$ ) is used to label the energy levels associated with the  $1s^2 2s^2 2p^6 3s^2 3p^4 4s$  and the  $1s^2 2s^2 2p^6 3s^2 3p^4 4p$  configurations.

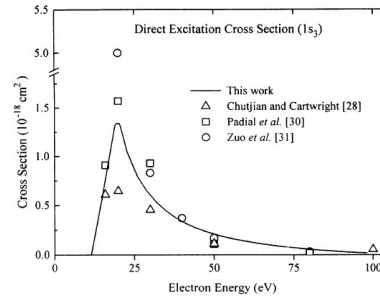
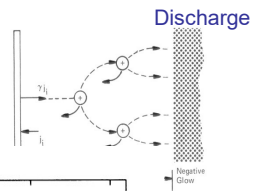


FIG. 13. Comparison of experimental and theoretical results for the energy dependence and magnitude of the direct excitation cross section for the  $1s_3 (J=0)$  metastable level.



PHYSICAL REVIEW A VOLUME 50, NUMBER 1 JULY 1994  
 Measurements of cross sections for electron-impact excitation into the metastable levels of argon and number densities of metastable argon atoms

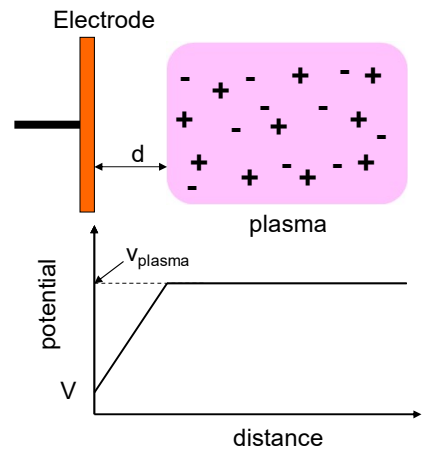
R. Scott Schappe, M. Bruce Schulman,\* L. W. Anderson, and Chun C. Lin  
 Department of Physics, University of Wisconsin, Madison, Wisconsin 53706  
 (Received 20 December 1993)

## Sheath width between the plasma and a negatively biased electrode (target or substrate)

Child-Langmuir formula

$$d^2 = \frac{4\epsilon_0}{9} \left( \frac{2e}{m} \right)^{1/2} \frac{V^{3/2}}{j}$$

$V$  – bias voltage  
 $j$  – ion current density  
 $m$  – ion mass



for  $Ar^+ \rightarrow d^2 [m^2] = \frac{4 \times 8.854 \times 10^{-12}}{9} \left( \frac{2 \times 1.6 \times 10^{-19}}{40 \times 1.66 \times 10^{-27}} \right)^{1/2} \frac{V^{3/2} [V]}{j \left[ \frac{A}{m^2} \right]} = 8.64 \times 10^{-9} \frac{V^{3/2}}{j}$

Mean free path  
 $\lambda = \frac{1}{\sqrt{2} n \sigma}$

We use this expression to compare the sheath width with the ion mean free path and evaluate the energy of the impinging ions on the electrode (target or substrate)

## Plasma parameters

- Electron density & temperature :  $n_e, T_e$
- Ion density & temperature :  $n_i, T_i$
- Plasma potential :  $V_{\text{plasma}}$

## - Assumptions

- Singly charged ions :  $n_e \sim n_i$
- Cold ions :  $T_i \ll T_e$

- For film growth we need ion flux  $J_i$  and ion energy

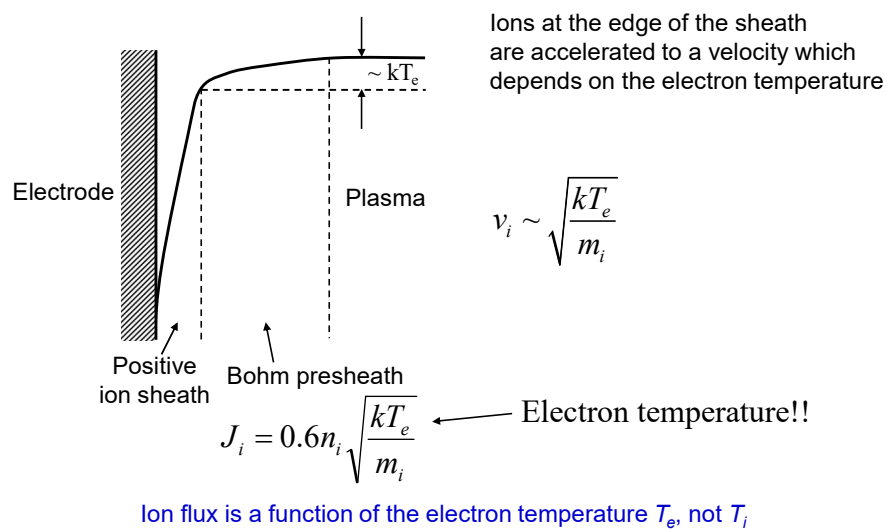
$$E_i \sim e(V_{\text{substrate}} - V_{\text{plasma}})$$

- Fluxes (in analogy with the kinetic theory of gases)

$$J_{i,e} = \frac{n_{i,e} \overline{v_{i,e}}}{4} = \frac{n_{i,e}}{4} \sqrt{\frac{8kT_{i,e}}{\pi m_{i,e}}} = 0.4n_{i,e} \sqrt{\frac{kT_{i,e}}{m_{i,e}}}$$

19

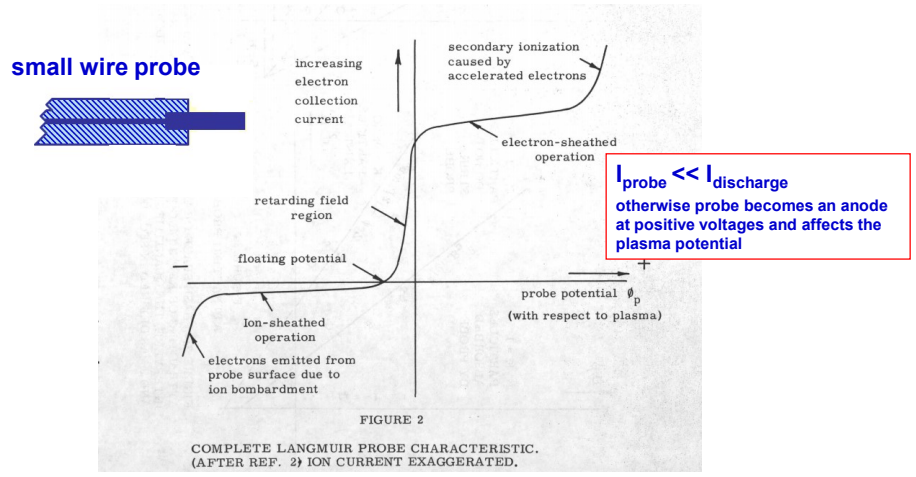
## Bohm presheath



S D Baalrud and C C Hegna, Kinetic theory of the presheath and the Bohm criterion, Plasma Sources Sci. Technol. 20 (2011) 025013

20

# Langmuir Probe Characteristics

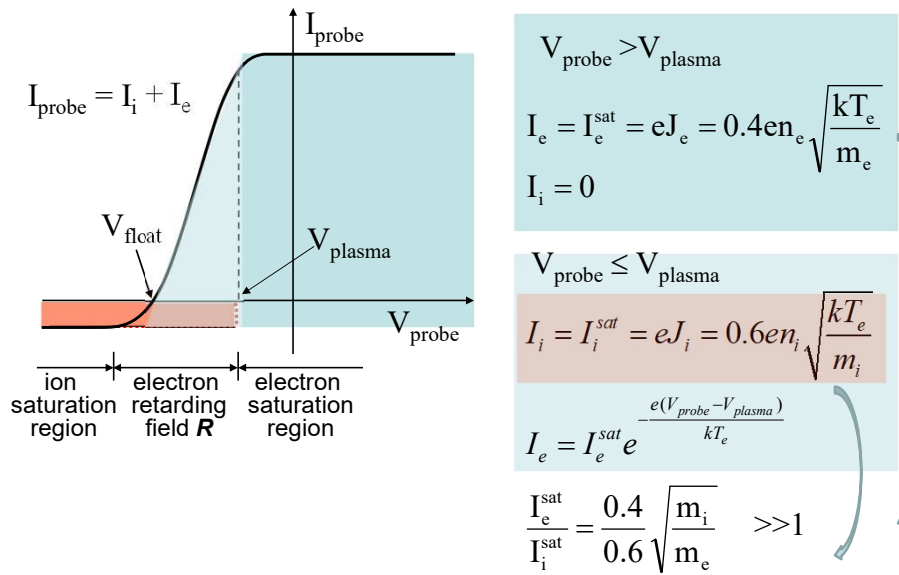


Laframboise, J.G.  
University of Toronto, Institute of Aerospace Studies, Report No 100

Available to download - Google it.

21

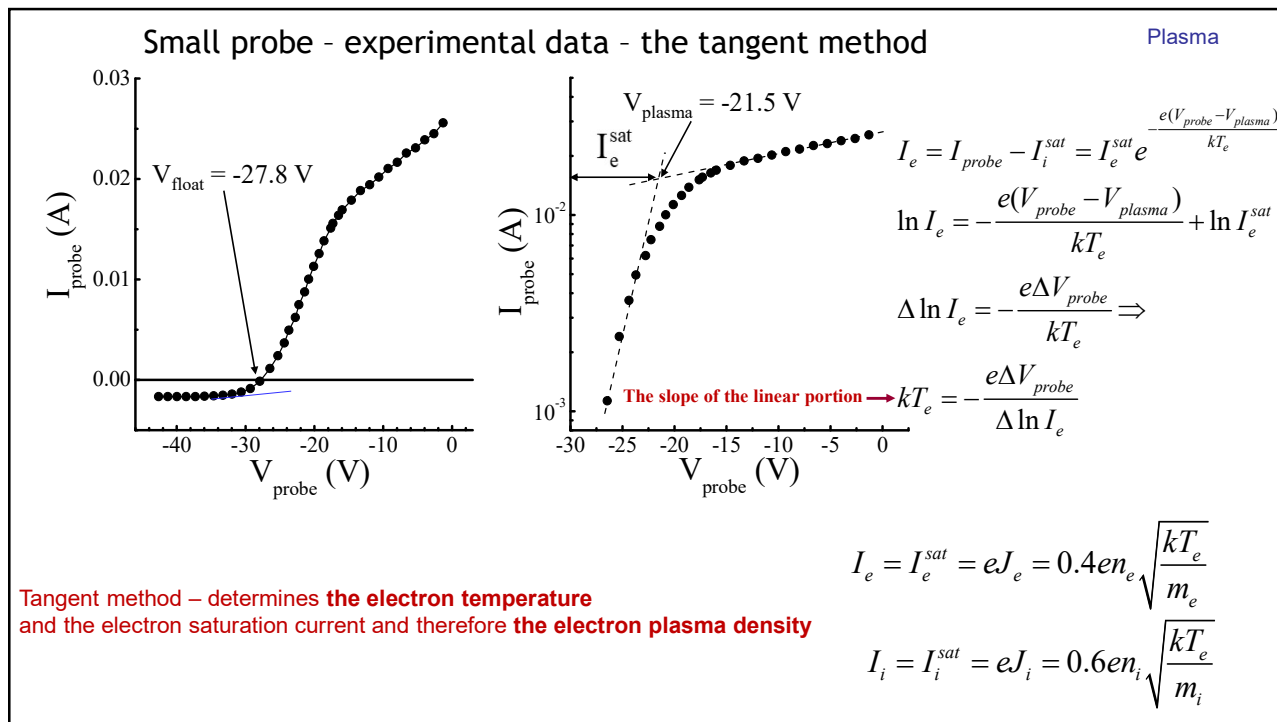
# Idealized Probe Characteristics



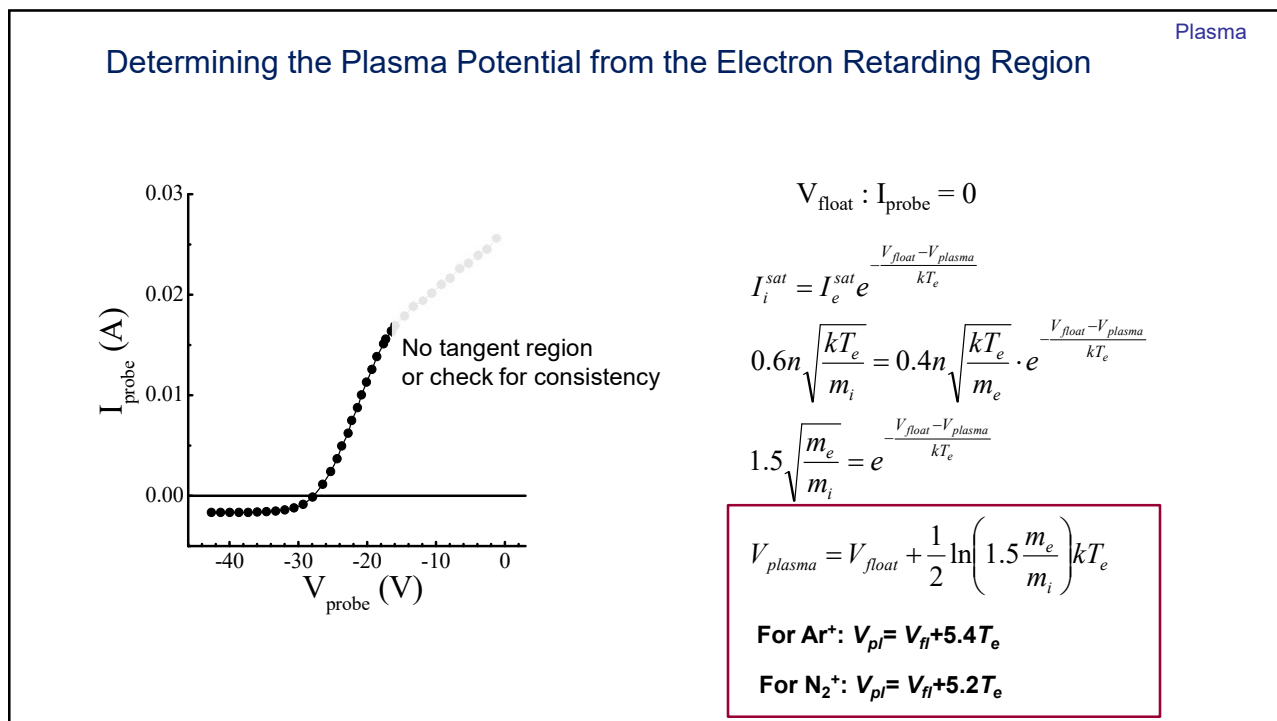
**Electrons much lighter; for Ar<sup>+</sup> the ratio is ~ 200**  
NB; magnetic field affects the motion of the electrons and this ratio could be smaller

22

22

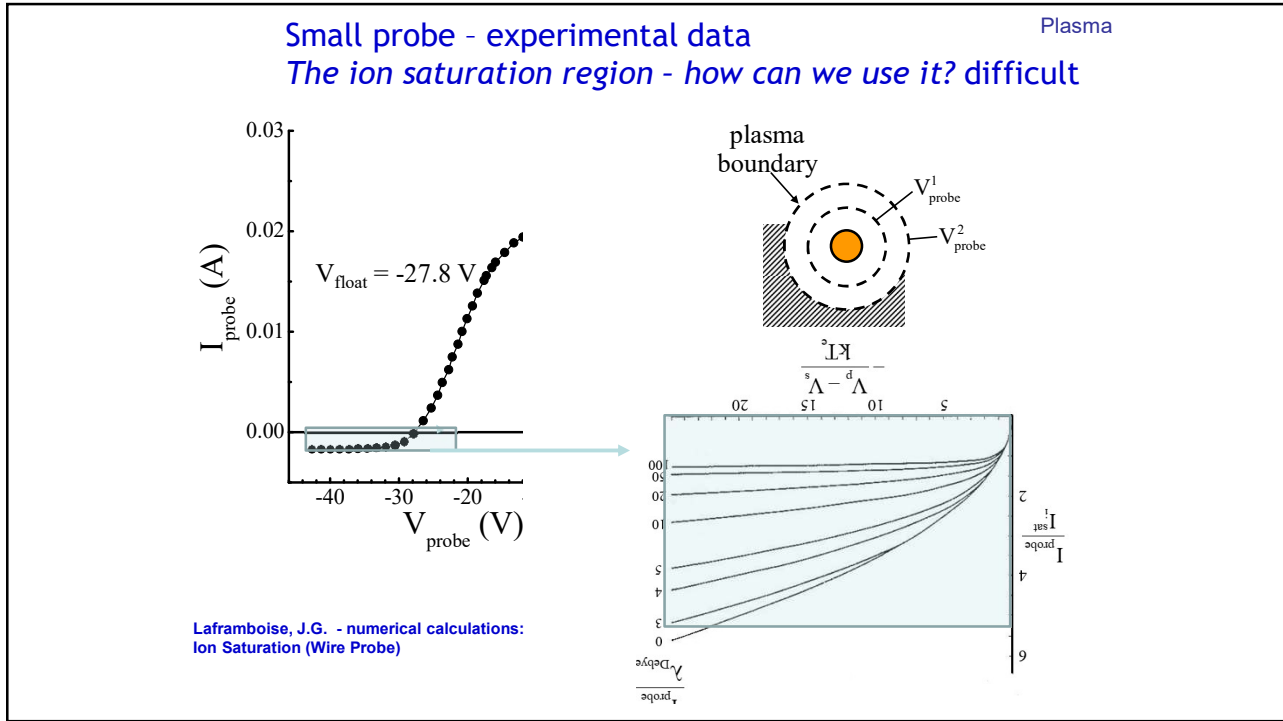


23

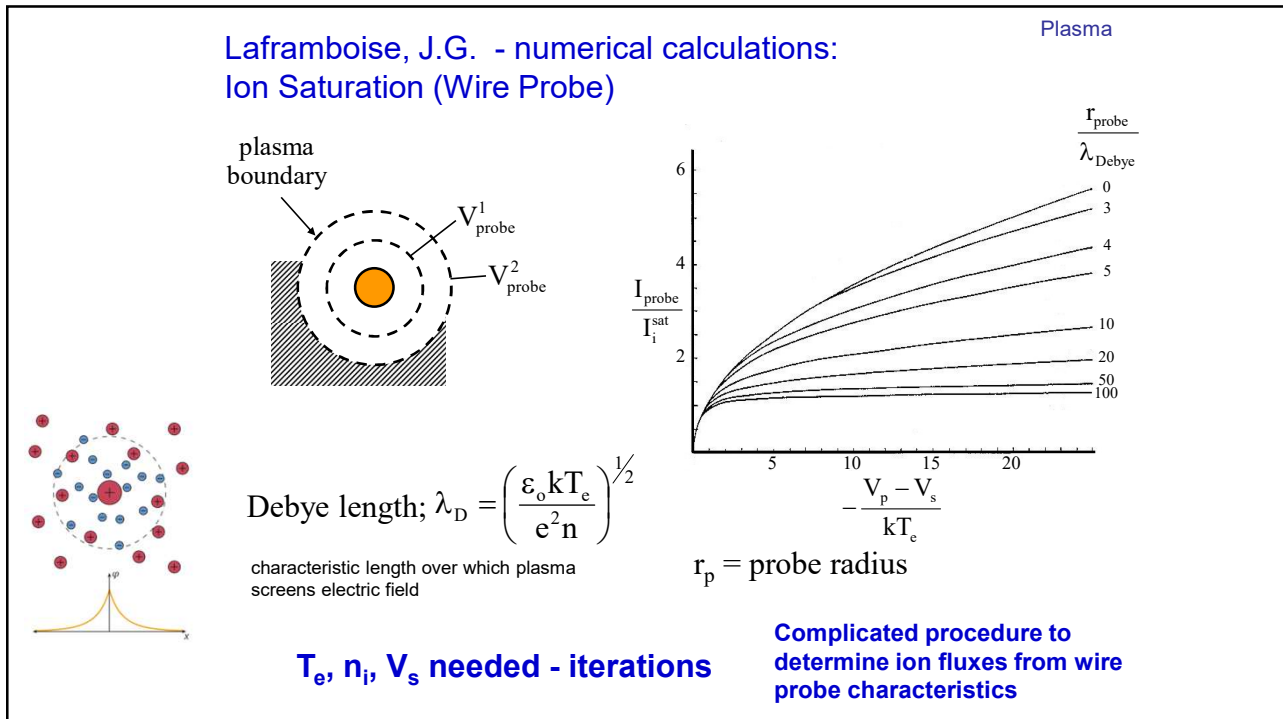


24





25



26

### Experimental Set-up for Flat Probe

(a) Vacuum chamber with the magnetron assembly and the substrate holder

(b) Electrostatic flat probe – embedded in the substrate holder at the same plane with minimal spacing  $\sim 1/4$  mm

(c) Apply the same potential to the probe and the substrate holder

I. Petrov et al *Contrib. Plasma Phys.*, 30 223 (1990).

27

### Flat Probe, best to measure the ion saturation current

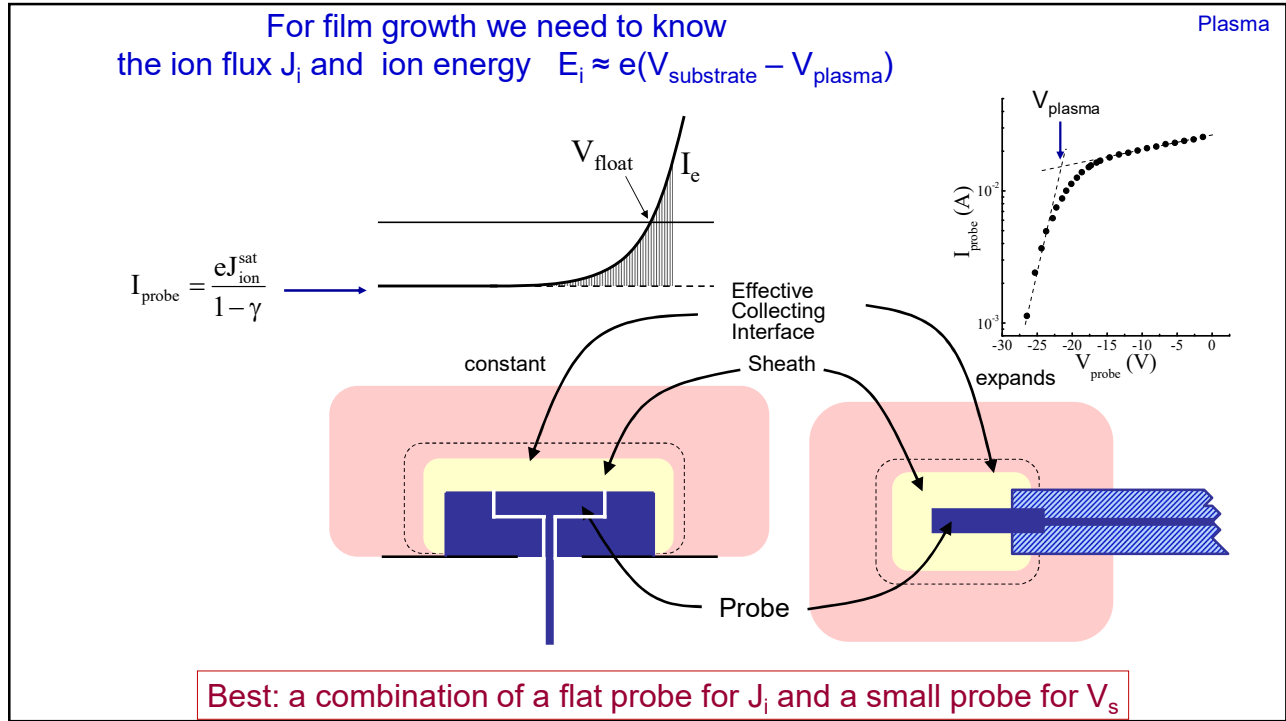
Plasma

- Directly measure ion flux; 
$$I_{\text{probe}} = \frac{eJ_{\text{ion}}^{\text{sat}}}{1 - \gamma}$$
  - $\gamma$  : Secondary electron emission coefficient
  - $\gamma$  correction of the order  $\sim 10\%$
- Big probe
  - Electron saturation disturbs the plasma (**acts as an anode**).
  - Tangent method cannot be used.
  - Plasma potential and electron temperature can still be estimated

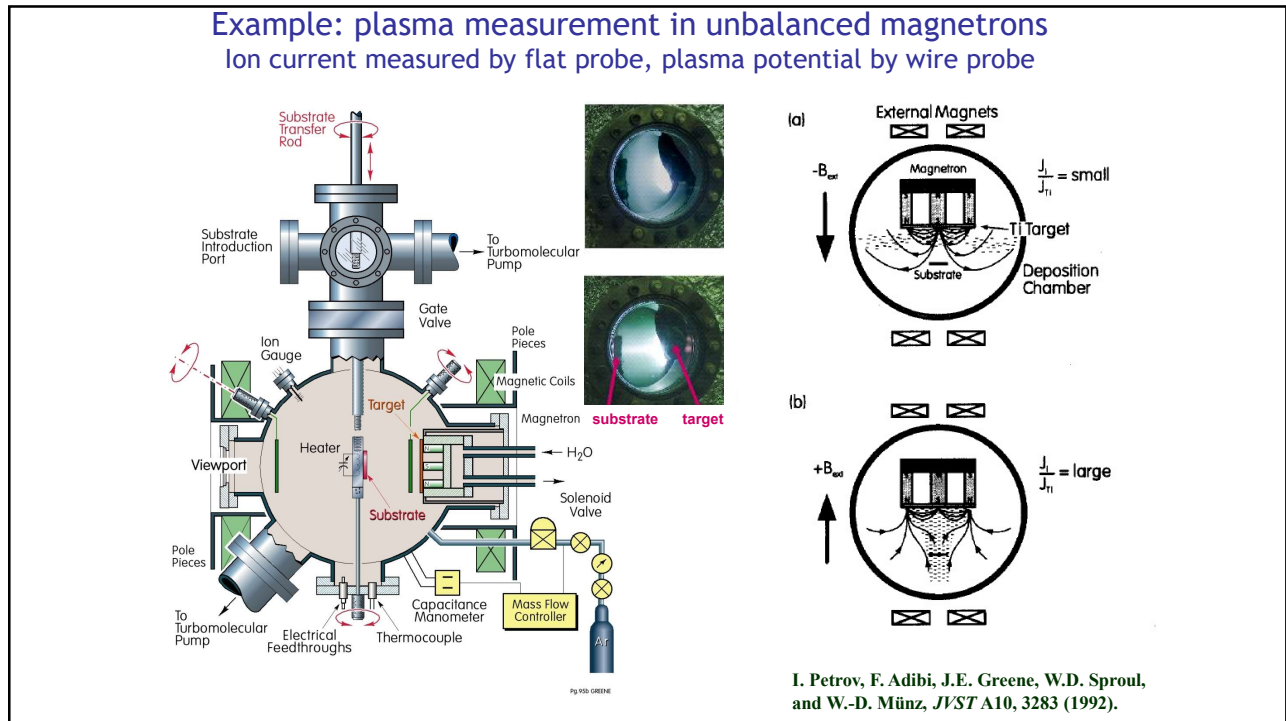
$T_e$  can be determined from the initial portion of the exponential rise of  $I_e$

$$V_{\text{plasma}} = V_{\text{float}} + \text{const. } T_e$$

28



29

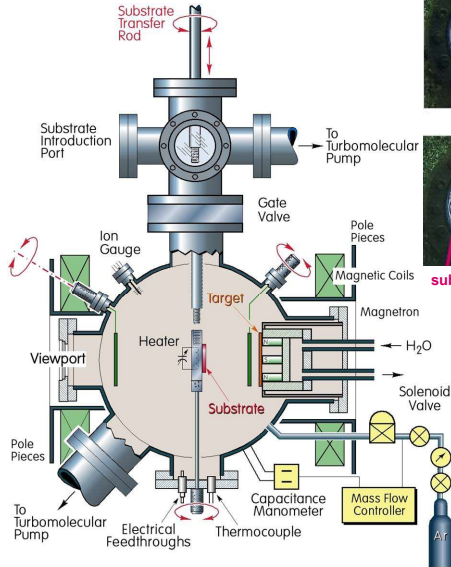


30

Example: plasma measurement in unbalanced magnetrons  
 Ion current measured by flat probe, plasma potential by wire probe

Plasma

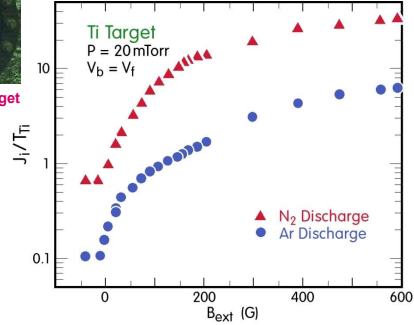
Unbalancing by external coils



$$E_i = e(V_{\text{plasma}} - V_{\text{bias}})$$

$$J_i = f(B_{\text{ext}})$$

$J_{Ti}$  remains approximately constant as a function of  $B_{\text{ext}}$ .



I. Petrov, F. Adibi, J.E. Greene, W.D. Sproul, and W.-D. Münz, *JVST A10*, 3283 (1992).

31

Plasma potential and density in unbalanced magnetrons (type II)  
 depends on whether the substrate is grounded or not

Plasma

Floating or (-) bias

Grounded substrate

I. Petrov, F. Adibi, J.E. Greene, W.D. Sproul, and W.-D. Münz, *JVST A10*, 3283 (1992).

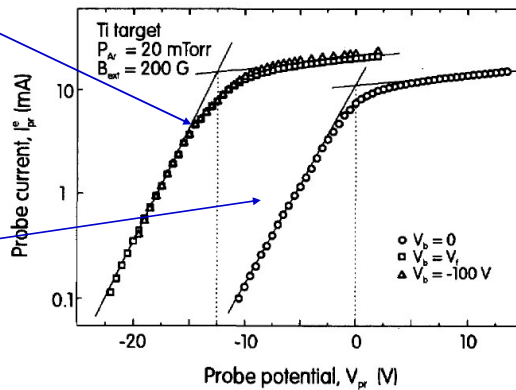


FIG. 7. Probe current  $I_{pr}^e$  due to electrons vs the applied probe potential  $V_{pr}$  during sputtering of Ti in a 20 mTorr Ar discharge with  $B_{\text{ext}} = 200$  G and the applied substrate bias  $V_b = 0$ ,  $V_f$ , and  $-100$  V. The plasma potential  $V_p$  (dotted lines) is 0 for  $V_b = 0$  and  $-12.5$  V for  $V_b = V_f$  and  $-100$  V.



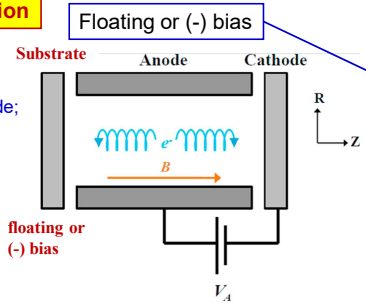
32

Plasma potential and density in unbalanced magnetrons (type II) depends on whether the substrate is grounded or not

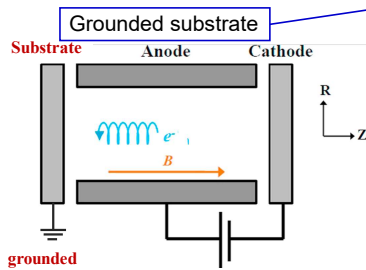
Plasma

Penning-type configuration

The substrate is NOT an anode; Collects no/few electrons



The substrate is the anode; collects all electrons



I. Petrov, F. Adibi, J.E. Greene, W.D. Sproul, and W.-D. Münz, *JVST A10*, 3283 (1992).

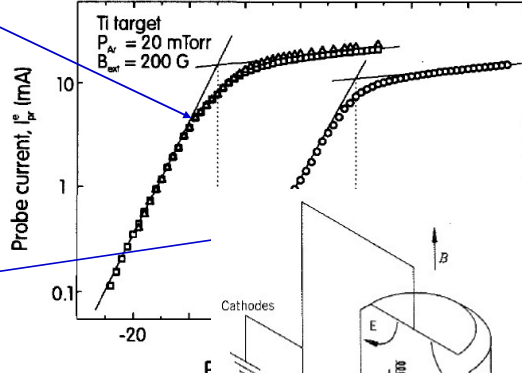


FIG. 7. Probe current  $I_{pr}^e$  du  $V_p$  during sputtering of Ti in and the applied substrate b potential  $V_p$  (dotted lines) is  $-100$  V.

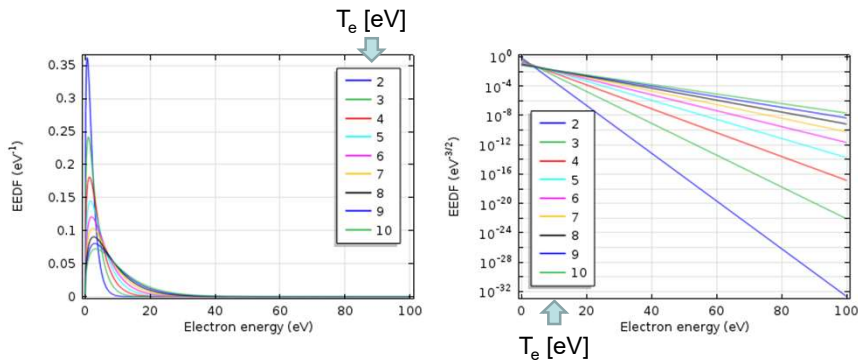
NB significant heating by electrons – measure substrate temperature rise

33

More on the Electron Energy Distribution Functions

Plasma

Maxwell electron energy distribution functions  
Elastic collision between electrons dominate



$$f(\epsilon) = \frac{2}{(kT)^{3/2}} \sqrt{\frac{\epsilon}{\pi}} \exp\left(\frac{-\epsilon}{kT}\right)$$

34

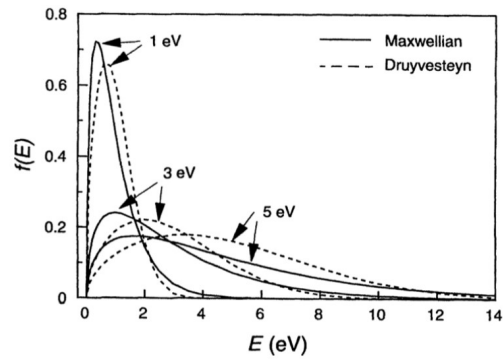
34

**Druyvesteyn EEDF**  
Elastic collision between electrons and cold atoms included

**Maxwell EEDF**  
Elastic collision between electrons dominate

$$f_D(\epsilon) = \frac{0.5648n_e}{(kT)^{3/2}} \sqrt{\epsilon} \exp \left[ -0.243 \left( \frac{\epsilon}{kT} \right)^2 \right]$$

$$f(\epsilon) = \frac{2}{(kT)^{3/2}} \sqrt{\frac{\epsilon}{\pi}} \exp \left( \frac{-\epsilon}{kT} \right)$$

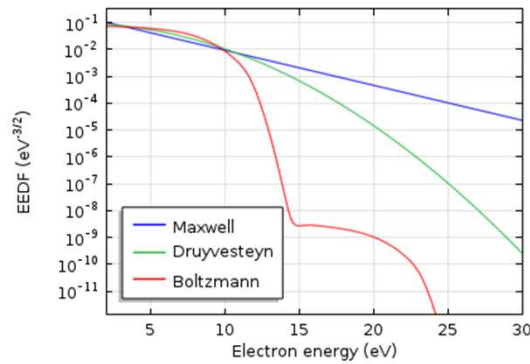


35

A. Grill, Cold Plasma Materials Fabrication: From Fundamentals to Applications. Wiley, 1994.

35

**Boltzmann EEDF includes inelastic collisions**



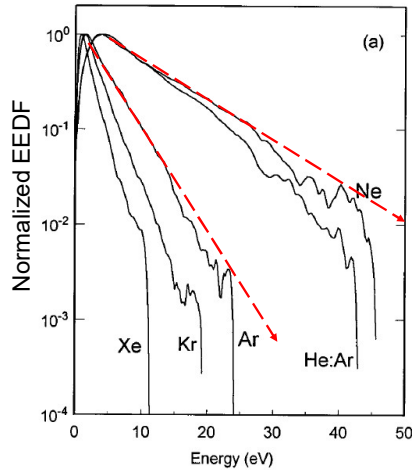
Computational example from:

<https://www.comsol.com/>

36

36

Measured Electron Energy Distribution Functions

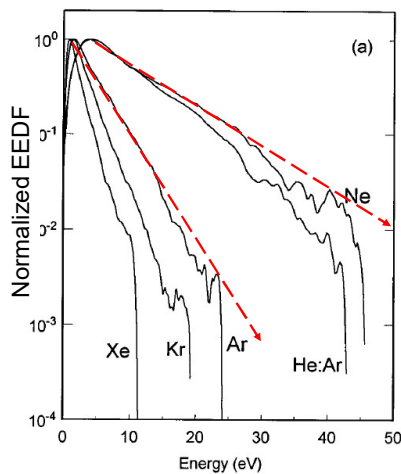


- The linear portion corresponds to Maxwell-like energy distribution
- Electron temperature  $T_e$  is the slope of the linear portion of the curve
- $T_e$  decreases as the ionization potential of the inert gas decreases
- In the area of inelastic collisions the high-energy tail is underpopulated
- extrapolation of the EEDF from the elastic energy range into the inelastic energy range may cause a significant error in the calculation of excitation and ionization rates
- **Metal dominated plasmas in HIPIMS lack high energy electron needed to ionize the inert gas.**

37  
A. Schwabedissen et al Physical review E, 55 (1997) 3450

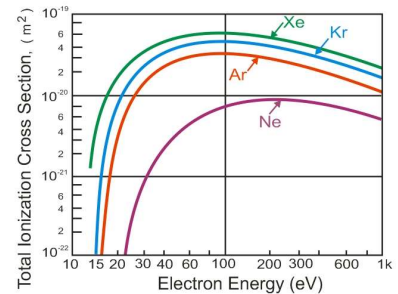
37

Measured Electron Energy Distribution Functions

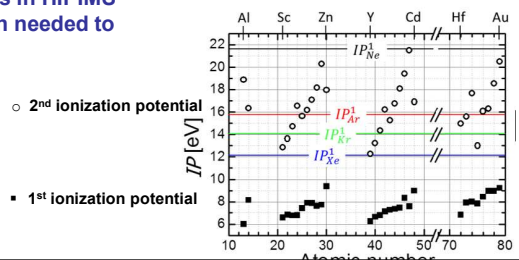


- The linear portion corresponds to Maxwell-like energy distribution
- Electron temperature  $T_e$  is the slope of the linear portion of the curve
- $T_e$  decreases as the ionization potential of the inert gas decreases
- In the area of inelastic collisions the high-energy tail is underpopulated
- extrapolation of the EEDF from the elastic energy range into the inelastic energy range may cause a significant error in the calculation of excitation and ionization rates
- **Metal dominated plasmas in HIPIMS lack high energy electron needed to ionize the inert gas.**

Electron impact ionization



Ionization potentials



38  
A. Schwabedissen et al Physical review E, 55 (1997) 3450

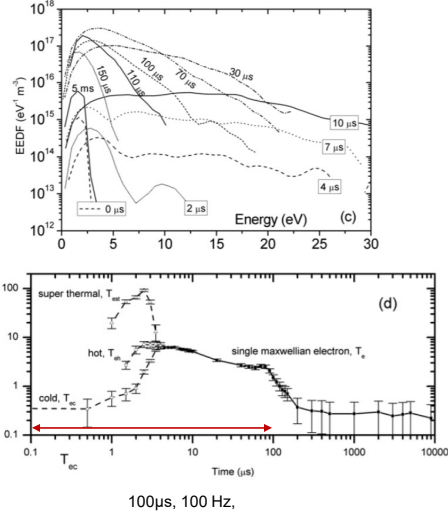
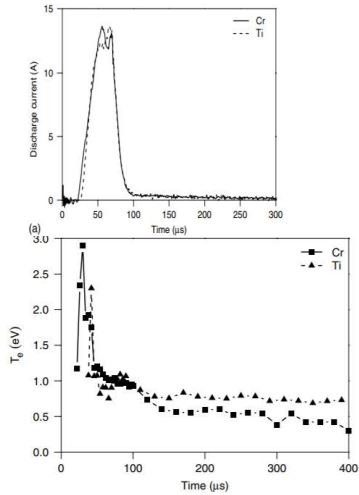
38

## Cooling of the electrons in metal-rich phase of HIPIMS discharges

Plasma

Alena Vetushka and Arutiun P Ehasarian  
*J. Phys. D: Appl. Phys.* 41 (2008) 015204

P Poolcharuansin and J W Bradley  
*Plasma Sources Sci. Technol.* 19 (2010) 025010



**Figure 6.** Effective electron temperature as a function of time for Cr and Ti targets at a pressure of 0.28 Pa.

39

## Plasmas: practical example

Plasma

$$d^2 [\text{cm}] = 8.64 \times 10^{-6} \frac{V^{3/2} [\text{V}]}{j \left[ \frac{\text{mA}}{\text{cm}^2} \right]}$$

$$\lambda_{CE} = \frac{kT}{p [\text{Pa}] \sigma [\text{m}^2] \sqrt{2}}$$

$$\sigma_{CE}^{Ar} = 4 \times 10^{-19} \left[ \text{m}^2 \right] \quad \text{Cross-section for charge exchange collisions for Ar}$$

Sputtering system		Voltage	Current density	Sheath width	$\lambda_{Ar}$
Diode $p_{Ar} = 70$ mTorr	target	3000 V	1 mA cm <sup>-2</sup>	12 mm	0.7 mm
	substrate	100 V	0.1 mA cm <sup>-2</sup>	3 mm	0.7 mm
magnetron $p_{Ar} = 3$ mTorr	target	500 V	50 mA cm <sup>-2</sup>	0.5 mm	16 mm
	substrate	100 V	1 mA cm <sup>-2</sup>	0.9 mm	16 mm

For magnetron sputtering  $\lambda_{Ar} >$  sheath width  $\rightarrow E_i \approx e(V_{\text{electrode}} - V_{\text{plasma}})$

40

40



## Cold Cathode Discharge

### Ion surface interactions

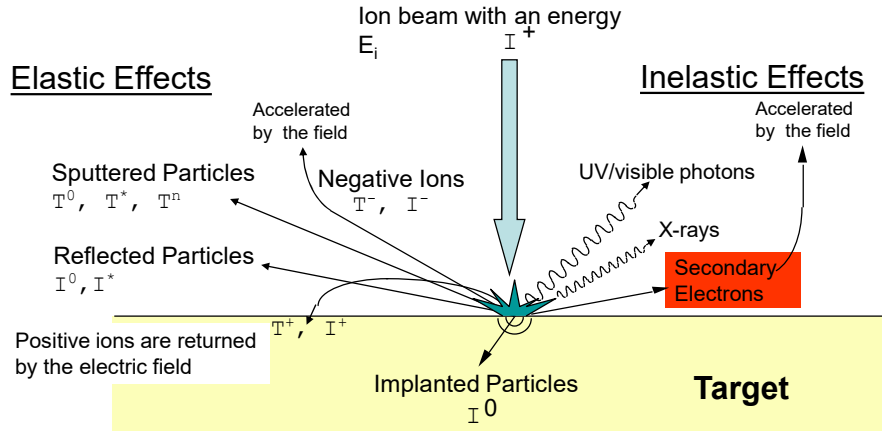


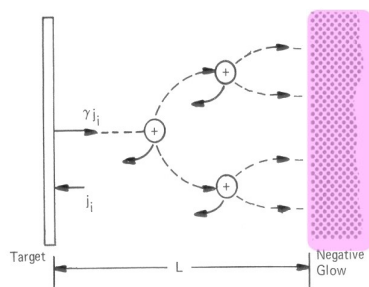
Figure after G.M. McCracken, Rep. Prog Phys. **28**, 241 (1975).

41

41

## Glow Discharge Voltage

Discharge



On the average, a secondary electron emitted from the cathode must acquire energy sufficient to produce a number of ions to release one further electron from the cathode.

$$\gamma_i \frac{V_T}{E_o} = 1$$

$$\frac{V_T}{E_o} = \frac{1}{\gamma_i \epsilon_i \epsilon_e}$$

$E_o$  – the average energy to produce an ion (for Ar  $\sim 30$  eV)

$\gamma_i$  – the secondary ion-electron yield (for Ar  $\sim 0.1$ )

$\epsilon_i$  – ion collection efficiency (fraction of created ions that strike the target)

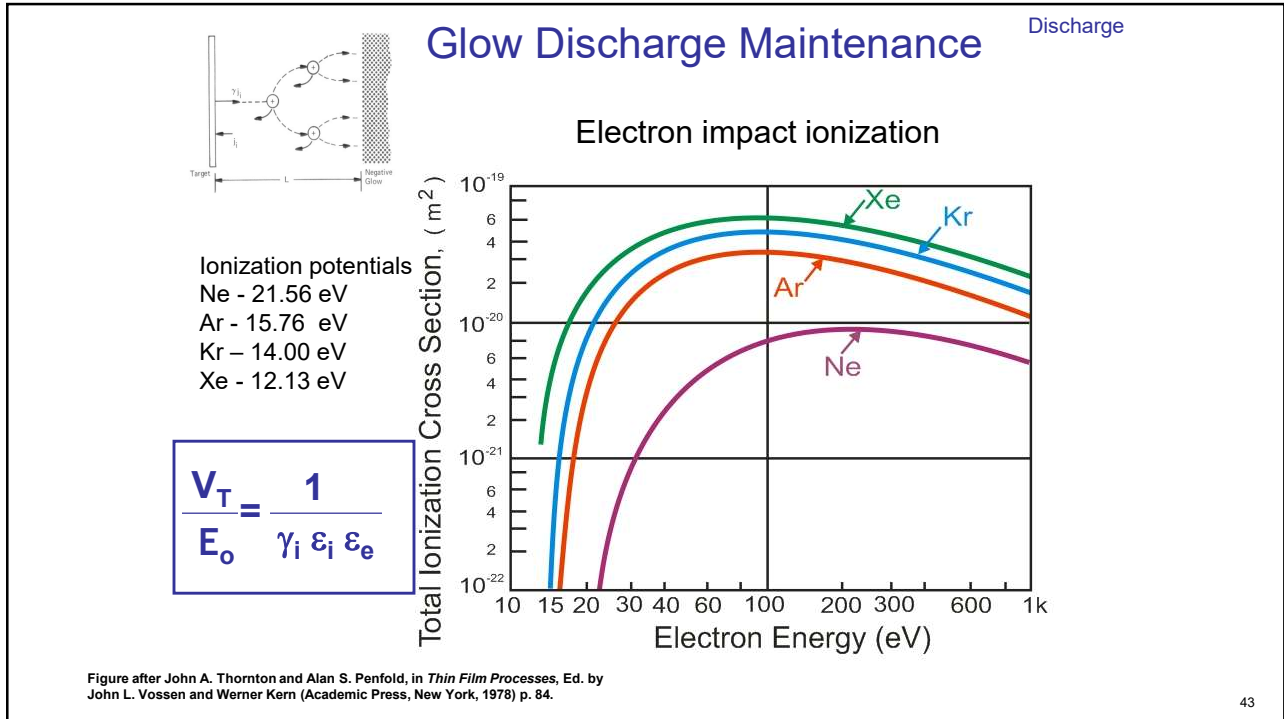
$\epsilon_e$  – fraction of the secondary electrons participating in ionization

$\epsilon_i \epsilon_e$  – characterizes the effectiveness of the ionization system; for magnetrons  $\sim 1$

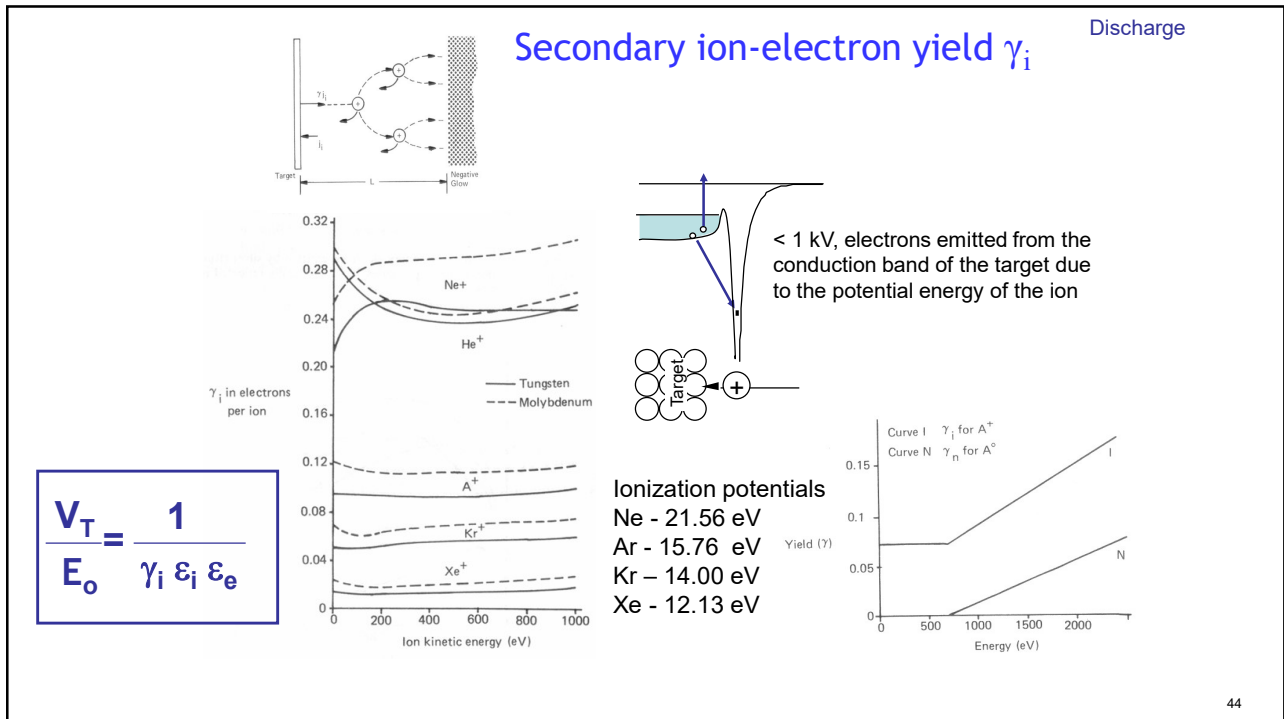
John A. Thornton and Alan S. Penfold, in *Thin Film Processes*, Ed. by John L. Vossen and Werner Kern (Academic Press, New York, 1978) p. 86.

42

42



43

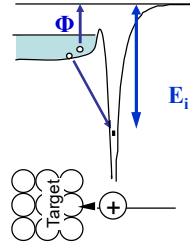
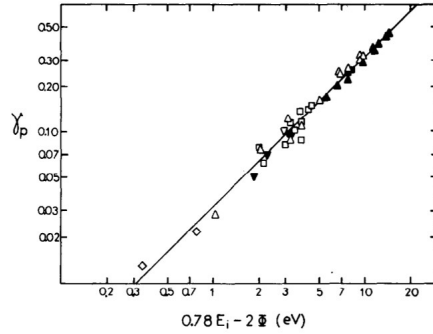


44

**ION-INDUCED ELECTRON EMISSION FROM CLEAN METALS \***

Surface Science 90 (1979) 240–255

R.A. BARAGIOLA, E.V. ALONSO, J. FERRON and A. OLIVA-FLORIO  
 Centro Atómico Bariloche, Comisión Nacional de Energía Atómica, Instituto Balseiro,  
 Un



$$V_T = \frac{E_o}{\gamma_i \epsilon_i \epsilon_e}$$

Note on reactive sputtering

Fig. 1. PEE yields  $\gamma_p$  versus  $0.78E_i - 2\Phi$ ; ( $\blacktriangle, \triangle$ ) Arifov [6] for  $Ne^+$  and  $Ar^+$  ions respectively; ( $\bullet, \circ, \nabla, \diamond, \blacksquare$ ) Hagstrum [7] for  $He^+, Ne^+, Ar^+, Kr^+$  and  $Xe^+$  ions respectively; ( $\square$ ) Oechsner [9] for  $Ar^+$  ions. The line is a least-square fit.

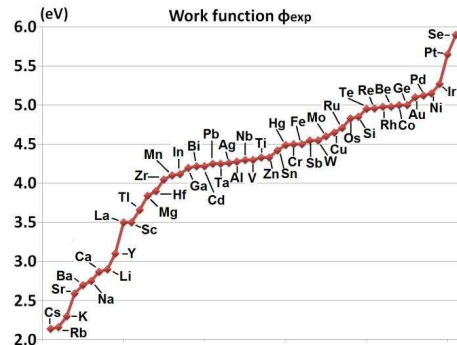
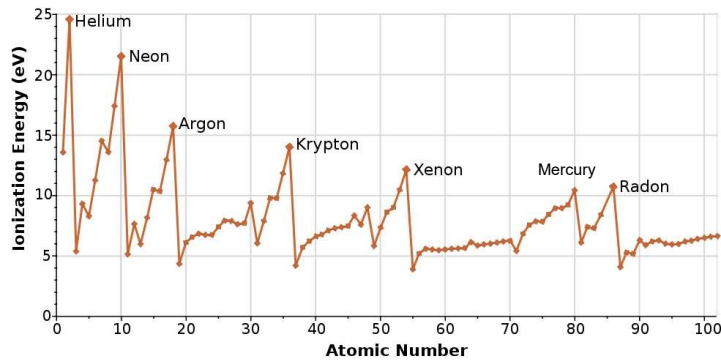
$\gamma = 0.032(0.78E_i - 2\Phi) \longrightarrow$  Note: potential electron emission possible for  $E_i > 2.56 \Phi$

- $\gamma$  – secondary ion-electron emission coefficient
- $E_i$  – ionization potential of the ion
- $\Phi$  – workfunction of the material

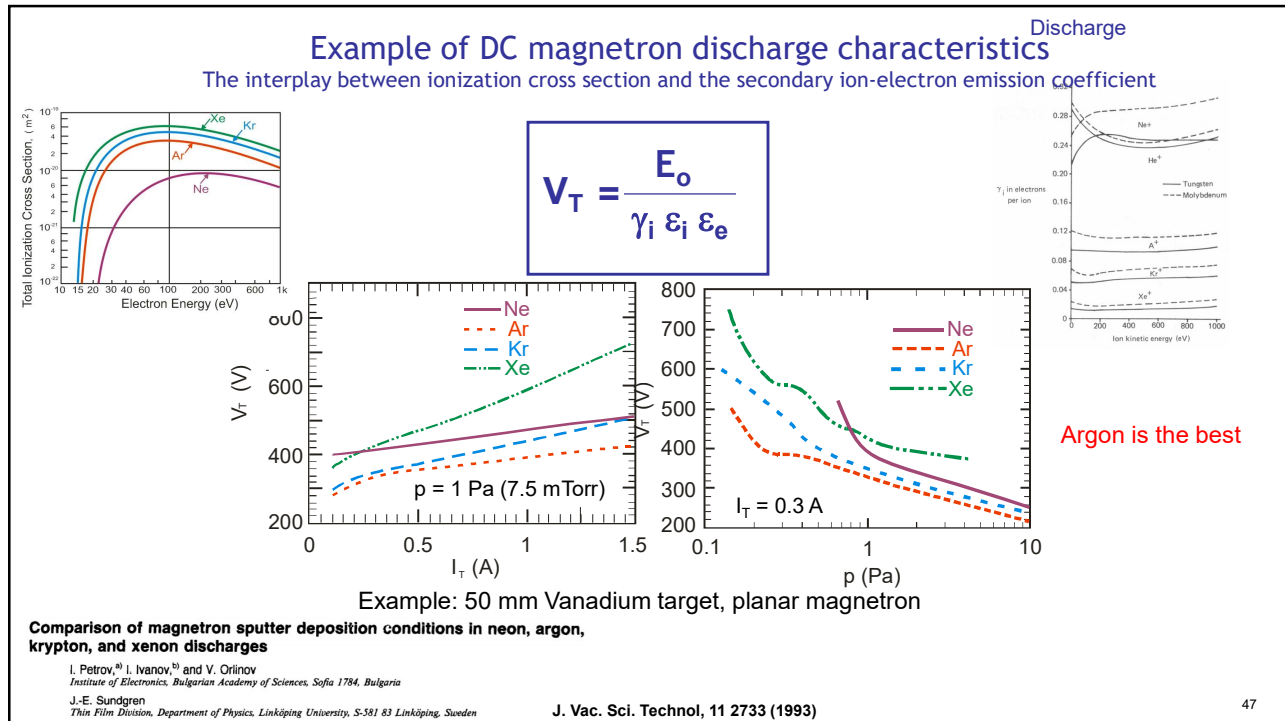
Note: potential electron emission possible for  $E_i > 2.56 \Phi$

For most metal ions  $E_i < 2.56 \Phi$

Sputtering discharges in pure metal plasmas (with singly charged ions) generally not possible  
 Reported for Cu under special conditions  
 Inter gas recycling in HiPIMS during the metal ion dominated phase



Ebong et al  
 DOI: 10.1109/HONET.2012.6421444



47



48

## Ion surface interactions

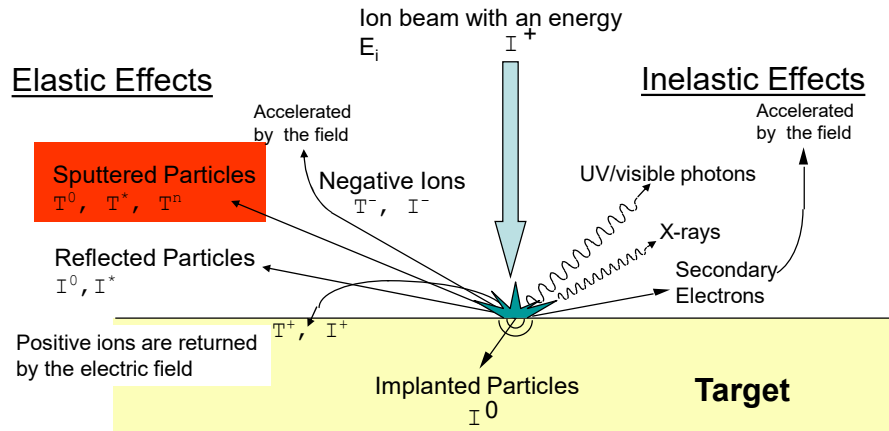
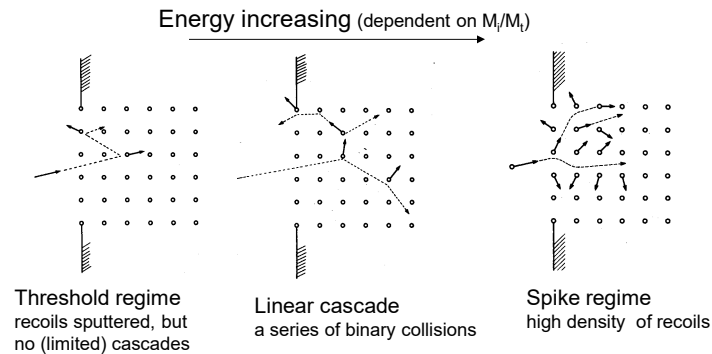


Figure after G.M. McCracken, Rep. Prog Phys. **28**, 241 (1975).

49

49

## Collision Cascade



- Ions striking a surface interact with a number of atoms in a series collisions.
- recoiled target atoms in turn collide with atom at rest generating a collision cascade.
- The initial ion energy and momentum are distributed to among the target recoil atoms.
- When  $E_i > 1$  keV, the cascade is "linear", i.e. approximated by a series of binary collisions in a stationary matrix.

P.Sigmund, "Sputtering by ion bombardment: theoretical concepts", in *Sputtering by particle bombardment I*, edited by R. Behrish, Springer-Verlag, 1981

50

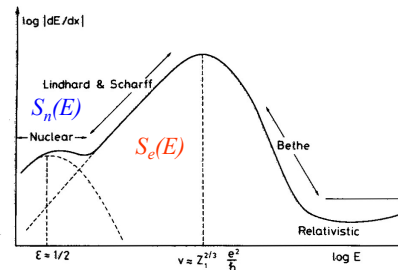
50

## Stopping cross section

$$\frac{dE}{dx} = -NS(E) = -N[S_n(E) + S_e(E)]$$

$S_n(E)$  - nuclear stopping  
 $S_e(E)$  - electronic stopping

$$Range = \int_0^E \frac{dE'}{NS(E')}$$



P.Sigmund, "Sputtering by ion bombardment: theoretical concepts", in *Sputtering by particle bombardment I*, edited by R. Behrish, Springer-Verlag, 1981

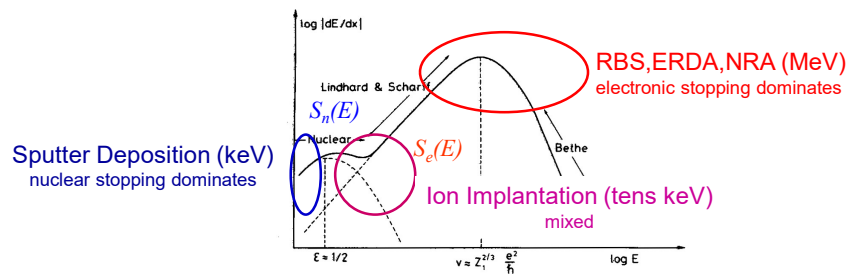
51

51

## Stopping cross section

$$\frac{dE}{dx} = -NS(E) = -N[S_n(E) + S_e(E)]$$

$$Range = \int_0^E \frac{dE'}{NS(E')}$$



P.Sigmund, "Sputtering by ion bombardment: theoretical concepts", in *Sputtering by particle bombardment I*, edited by R. Behrish, Springer-Verlag, 1981

52

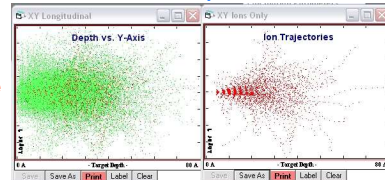
52

# SRIM/TRIM simulation

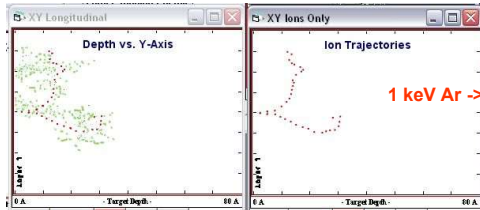
Monte-Carlo simulation of ion implantation, reflection, recoil cascades, and sputtering

1 keV Ar -> Be

200 impacts



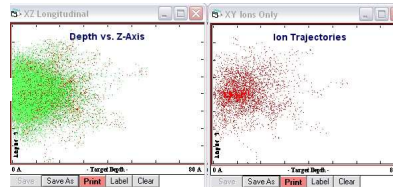
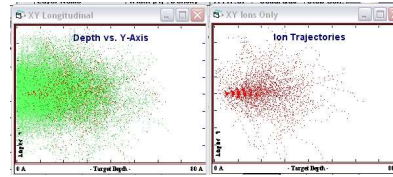
2 impacts



1 keV Ar -> Ti

ion trajectories in red  
recoils in green

1 keV Ar -> W



James F. Ziegler, IBM : <http://www.srim.org/>

# Sputtering Yield (Y)

- Sputtering begins at an energy threshold and increases rapidly
- Then nearly linearly**
- As the energy increases the curve levels off.

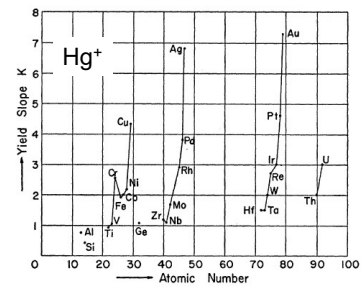
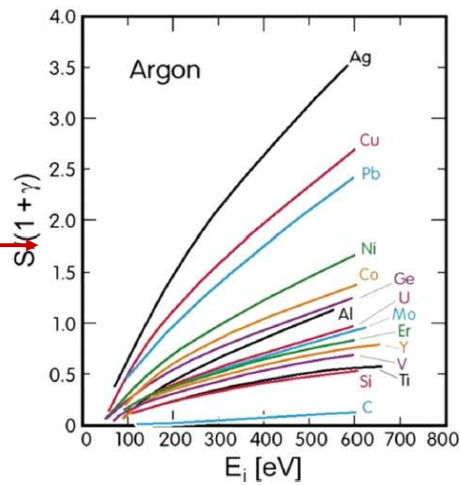


FIG. 7. Yield slopes vs atomic number.  
Slope of linear portion vs atomic number

Data from R.Y. Stuart and G.K. Wehner, J. Appl. Phys. 33, 2351 (1962); G.K. Wehner, Phys Rev, 112, 1120, 1958  
D. Rosenberg and G.K. Wehner, J. Appl. Phys. 33, 1842 (1962);

Sputtering Yields at Very Low Bombarding Ion Energies\*



Gottfried Karl Wehner  
23 September 1910 (Dresden)  
13 June 1996 (Munich)

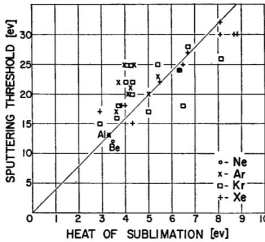
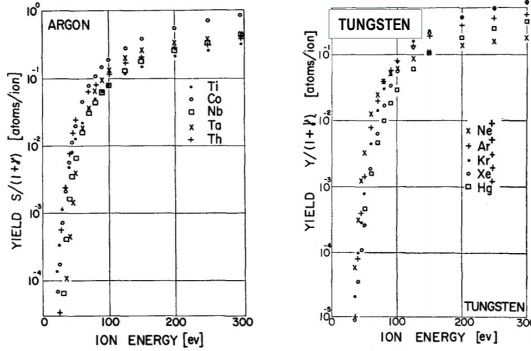


FIG. 13. Thresholds vs heats of sublimation.

TABLE II. Threshold energies. Circled values are those for which the energy transfer factor  $\frac{4m_1m_2}{(m_1+m_2)^2}$  is 0.9 or higher.

Be	Ne	Ar	Kr	Xe	Hg	Mo	24	24	24	Hg	32
Al	(12)	(15)	(15)	(18)	(18)	Rh	25	24	(28)	(28)	(27)
Ti	(12)	(15)	(15)	(18)	(18)	Pd	20	20	(28)	(28)	20
V	(12)	(15)	(15)	(18)	(18)	Ag	12	12	(17)	(17)	20
Cr	(12)	(15)	(15)	(18)	(18)	Ta	25	26	(30)	(30)	30
Fe	(12)	(15)	(15)	(18)	(18)	W	35	33	(30)	(30)	30
Co	(12)	(15)	(15)	(18)	(18)	Re	35	35	(25)	(25)	35
Ni	(12)	(15)	(15)	(18)	(18)	Pt	27	25	(22)	(22)	25
Cu	(12)	(15)	(15)	(18)	(18)	Au	20	20	(20)	(18)	25
Ge	(12)	(15)	(15)	(18)	(18)	Th	20	24	(25)	(25)	27
Zr	(12)	(15)	(15)	(18)	(18)	U	20	23	(25)	(25)	27
Nb	(12)	(15)	(15)	(18)	(18)						

Sputtering Yields of Metals for Ar<sup>+</sup> and Ne<sup>+</sup> Ions with Energies from 50 to 600 eV\*†

To this day:  
the most comprehensive set of  
low-energy sputtering yields data

NILS LAEGREID AND G. K. WEHNER  
Mechanical Division of General Mills, Incorporated, Minneapolis, Minnesota  
(August 24, 1960)

Sputtering yields for polycrystalline metal and semiconductor targets under normally incident Ar<sup>+</sup> and Ne<sup>+</sup> ion bombardment were measured in the energy range from 50 to 600 eV. The yields (atoms/ion) were determined by measuring the weight loss of spherical targets immersed like large negative Langmuir probes.

TABLE I. Sputtering yields for 28 elements under Ne<sup>+</sup> and Ar<sup>+</sup> ion bombardment.

Target	Neon				Yield at lowest ion energy		Argon			
	100 (ev)	200 (ev)	300 (ev)	600 (ev)	Y	E (ev)	100 (ev)	200 (ev)	300 (ev)	600 (ev)
Be	0.012	0.10	0.26	0.56	0.05	80	0.074	0.18	0.29	0.80
Al	0.031	0.24	0.43	0.83	0.11	100	0.11	0.35	0.65	1.24
Si	0.034	0.13	0.25	0.54	0.06	80	0.07	0.18	0.31	0.53
Ti	0.08	0.22	0.30	0.45	0.081	100	0.081	0.22	0.33	0.58
V	0.06	0.17	0.36	0.55	0.03	60	0.11	0.31	0.41	0.70
Cr	0.18	0.49	0.73	1.05	0.026	40	0.30	0.67	0.87	1.30
Fe	0.18	0.38	0.62	0.97	0.064	60	0.20	0.53	0.76	1.26
Co	0.084	0.41	0.64	0.99	0.048	60	0.15	0.57	0.81	1.36
Ni	0.22	0.46	0.65	1.34	0.067	60	0.28	0.66	0.95	1.52
Cu	0.26	0.84	1.20	2.00	0.10	60	0.48	1.10	1.59	2.30
Ge	0.12	0.32	0.48	0.82	0.017	30	0.22	0.50	0.74	1.22
Zr	0.054	0.17	0.27	0.42	0.027	60	0.12	0.28	0.41	0.75
Nb	0.051	0.16	0.23	0.42	0.017	60	0.068	0.25	0.40	0.65
Mo	0.10	0.24	0.34	0.54	0.027	60	0.13	0.40	0.58	0.93
Ru	0.078	0.26	0.38	0.67	0.012	60	0.14	0.41	0.68	1.30
Rh	0.081	0.36	0.52	0.77	0.19	100	0.19	0.55	0.86	1.46
Pd	0.14	0.59	0.82	1.32	0.033	50	0.42	1.00	1.41	2.39
Ag	0.27	1.00	1.30	1.98	0.22	60	0.63	1.58	2.20	3.40
Hf	0.057	0.15	0.22	0.39	0.004	40	0.16	0.35	0.48	0.83
Ta	0.056	0.13	0.18	0.30	0.01	60	0.10	0.28	0.41	0.62
W	0.038	0.13	0.18	0.32	0.008	60	0.068	0.29	0.40	0.62
Re	0.04	0.15	0.24	0.42	0.034	80	0.10	0.37	0.56	0.91
Os	0.032	0.16	0.24	0.41	0.057	100	0.057	0.36	0.56	0.95
Ir	0.069	0.21	0.30	0.46	0.019	60	0.12	0.43	0.70	1.17
Pt	0.12	0.31	0.44	0.70	0.032	60	0.20	0.63	0.95	1.56
Au	0.20	0.56	0.84	1.18	0.035	50	0.32	1.07	1.65	2.43 (500)
Th	0.028	0.11	0.17	0.36	0.017	60	0.097	0.27	0.42	0.66
U	0.063	0.20	0.30	0.52	0.14	100	0.14	0.35	0.59	0.97



Sputtering Yields for Low Energy He<sup>+</sup>, Kr<sup>+</sup>, and Xe<sup>+</sup>-Ion Bombardment\*

D. ROSENBERG AND G. K. WEHNER  
The General Mills Electronics Group, Minneapolis, Minnesota  
(Received November 2, 1961)

To this day:  
the most comprehensive set of  
low-energy sputtering yields data

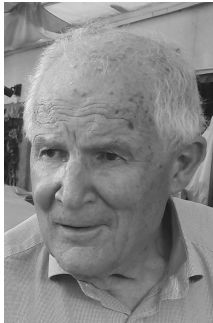
TABLE I. Sputtering yields for 30 elements under He<sup>+</sup>, Kr<sup>+</sup>, and Xe<sup>+</sup>-ion bombardment. Numbers in brackets are doubtful due to surface layers.

Target	Helium				Krypton				Xenon			
	100 (ev)	200 (ev)	300 (ev)	600 (ev)	100 (ev)	200 (ev)	300 (ev)	600 (ev)	100 (ev)	200 (ev)	300 (ev)	600 (ev)
Be	(0.040)	(0.095)	(0.15)	(0.34)	0.03	0.17	0.24	0.61	...	0.12	0.24	0.42
C	0.008	0.020	0.035	0.085	0.005	0.045	0.090	0.18	...	0.04	0.08	0.21
Al	...	0.005	0.008	0.021	0.09	0.30	0.52	1.11	0.06	0.24	0.45	1.02
Si	0.015	0.045	0.075	0.15	0.05	0.12	0.23	0.64	...	0.08	0.21	0.51
Ti	0.010	0.038	0.05	0.08	0.03	0.16	0.29	0.53	...	0.13	0.24	0.50
V	(0.003)	(0.020)	(0.038)	(0.09)	0.06	0.21	0.35	0.69	0.05	0.20	0.39	0.72
Cr	0.030	0.070	0.105	0.20	0.21	0.56	0.88	1.55	0.13	0.44	0.85	1.90
Mn	...	...	...	...	0.11	0.40	0.69	1.80	0.08	0.44	0.60	1.67
Fe	0.030	0.065	0.09	0.17	0.12	0.38	0.64	1.23	0.06	0.29	0.54	1.20
Co	0.010	0.042	0.075	0.15	0.08	0.30	0.50	1.33	0.09	0.38	0.61	1.30
Ni	0.028	0.060	0.095	0.18	0.16	0.47	0.75	1.50	0.10	0.37	0.71	1.48
Cu	0.045	0.11	0.16	0.27	0.33	0.92	1.42	2.80	0.26	0.79	1.29	2.44
Ge	0.010	0.03	0.05	0.08	0.12	0.37	0.66	1.35	0.08	0.31	0.54	1.20
Zr	...	(0.004)	(0.013)	(0.025)	0.04	0.18	0.34	0.72	0.03	0.18	0.31	0.71
Nb	...	0.005	0.010	0.030	0.03	0.17	0.30	0.68	0.02	0.17	0.31	0.61
Mo	(0.001)	(0.005)	0.015	0.040	0.07	0.32	0.54	1.05	0.06	0.28	0.51	1.06
Ru	...	...	...	...	0.08	0.45	0.77	1.45	0.05	0.37	0.71	1.42
Rh	0.004	0.015	0.030	0.065	0.16	0.54	0.90	1.70	0.12	0.51	0.84	1.60
Pd	0.020	0.057	0.082	0.16	0.33	0.97	1.47	2.55	0.34	0.93	1.39	2.48
Ag	0.030	0.082	0.125	0.23	0.40	1.35	1.85	3.90	0.40	1.05	1.80	4.20
Hf	...	...	(0.002)	(0.010)	0.12	0.39	0.59	1.02	...	0.30	0.55	1.17
Ta	...	0.002	0.003	0.012	0.07	0.33	0.53	0.98	0.05	0.32	0.50	1.00
W	...	0.001	0.004	0.008	0.06	0.36	0.58	1.07	0.03	0.35	0.60	1.18
Re	...	...	...	...	0.08	0.42	0.74	1.43	0.02	0.31	0.80	1.40
Os	(0.0005)	(0.002)	(0.004)	(0.022)	0.05	0.39	0.73	1.45	0.03	0.39	0.74	1.53
Ir	...	(0.001)	(0.003)	(0.010)	0.10	0.42	0.66	1.58	0.05	0.52	0.86	1.79
Pt	...	0.004	0.010	0.035	0.15	0.67	1.15	2.11	0.19	0.72	1.25	2.23
Au	...	0.020	0.035	0.08	0.42	1.10	1.90	3.42	0.16	1.00	1.83	3.10
Th	...	(0.0005)	...	(0.002)	0.09	0.36	0.60	1.07	0.08	0.35	0.60	1.22
U	...	0.002	0.004	0.013	0.14	0.47	0.79	1.46	...	0.24	0.45	1.00

Theory of Sputtering. I. Sputtering Yield of Amorphous and Polycrystalline Targets\*

PETER SIGMUND†

Metallurgy Division, Argonne National Laboratory, Argonne, Illinois 60439



$$Y(E) = \frac{0.04}{U} \alpha (M_i/M_t) S_n(E)$$

$\alpha$  - dimensionless coefficient

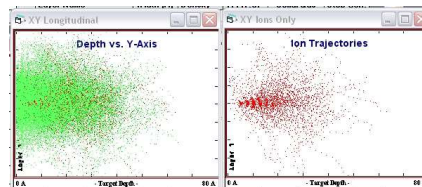
$\alpha$  - sublimation energy

$S_n(E)$ , nuclear stopping cross-section

$E$  - incident ion energy

Note: the sputtering yield is proportional to the nuclear stopping at the point of entry

**Peter Sigmund**  
University of Southern Denmark  
PhD. 1962, TU Aachen, under Prof. Guenter Leibfried  
1967-9 Visiting scientist, Argonne National Lab  
1965-7 Scientist, Research Center Jülich



## Sigmund's linear cascade formula for Y(E)

$$Y(E) = \frac{0.04}{U} \alpha(M_t/M_i) S_n(E)$$

$\alpha$  - dimensionless coefficient  
 $S_n(E)$ , collisional energy at the surface  
 (nuclear energy loss function)  
 $U$  - sublimation energy

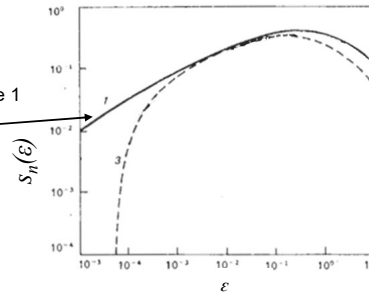
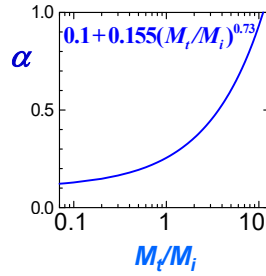
$$S_n(E) = 85 \frac{Z_i Z_t}{(Z_i^{2/3} + Z_t^{2/3})^{0.5}} \frac{M_i}{M_t + M_i} S_n(\epsilon)$$

$S_n(\epsilon)$  - function of the reduced energy which is the same for all ion-target combination

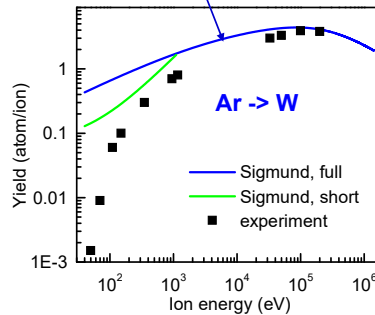
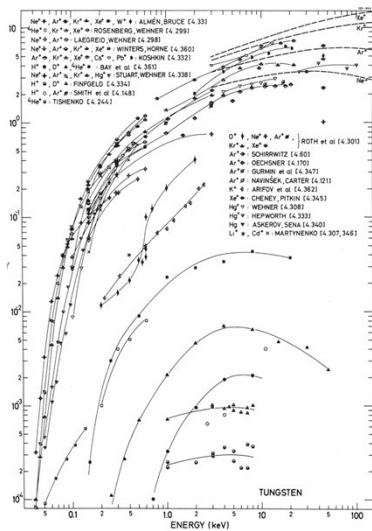
$$S_n(\epsilon) = \frac{3.441\sqrt{\epsilon} \ln(\epsilon + 2.718)}{1 + 6.355\sqrt{\epsilon} + \epsilon(6.882\sqrt{\epsilon} - 1.708)}$$

$\epsilon$  - reduced energy

$$\epsilon = \frac{0.03255}{Z_i Z_t (Z_i^{2/3} + Z_t^{2/3})^{0.5}} \frac{M_t}{M_t + M_i} E$$



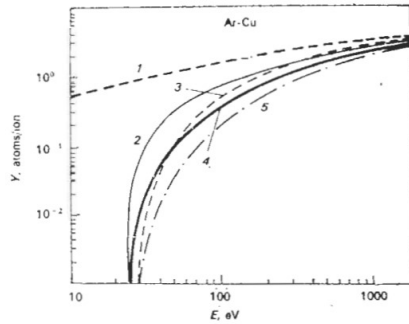
## Sigmund formula: good agreement in the 10s keV, poor in the < 1 keV



0.1 < E < 1kV, Sigmund derives a remarkably simple formula

$$Y(E) = \frac{3}{4\pi^2} \alpha(M_t/M_i) \frac{\gamma E}{U}$$

$$\gamma = \frac{4M_t M_i}{(M_t + M_i)^2}$$



$$Y(E) = Y_{\text{Sigmund}}(E) \cdot f(E),$$

$$f(E) = 0, E = E_{TH}$$

$$f(E) = 1, E \gg E_{TH}$$

Each formula comes with its choice of  $E_{TH}$

Fig. 2. Energy dependence of the sputtering yield: curve 1 — Sigmund's expression, curve 2 — Matsunami I, curve 3 — Bohdansky, curve 4 — Yamamura, curve 5 — Matsunami II

3. Matsunami, N., Y. Yamamura, Y. Itikawa, N. Itoh, Y. Kazumata, S. Miyagawa, K. Morita, R. Shimizu. — Rad. Eff. Letters, 57, 1980, p. 15.
4. Yamamura, Y., N. Matsunami, N. Itoh. — Rad. Eff., 71, 1983, p. 65.
5. Bohdansky, J. — Nucl. Instr. Meth., B2, 1984, p. 587.
6. Matsunami, N., Y. Yamamura, Y. Itikawa, N. Itoh, Y. Kazumata, S. Miyagawa, K. Morita, R. Shimizu, H. Tawara. — Atomic Data and Nucl. Data Tables, 31, 1984, p. 1.

Petrov, V. Orlinov, S. Grudeva, Comparison of the Low-Energy Sputtering Yield Formulas, *Bulg. J. Phys.*, 19 102 (1991).

61

**The best formula for low energy sputtering yield:  
Y. Yamamura et al formula**

Sputtering begins at an energy threshold that depends on the efficiency of momentum transfer to the target. This depends on the mass match. It also depends on the surface binding energy of atoms in the target.

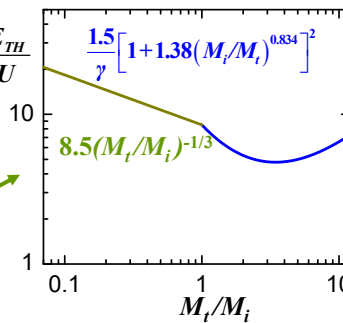
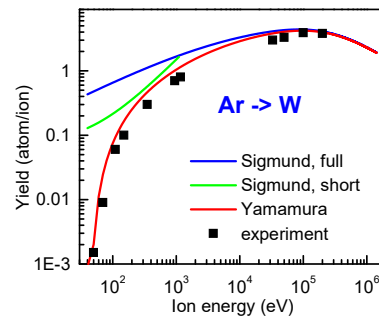
Petrov et al:

Y. Yamamura's correction for  $f(E)$ ,  $E_{TH}$  is best for Sputter-deposition combinations ion/target

$$Y(E) = \frac{0.04}{U} \alpha(M_t / M_i) Q S_n(E) \left( 1 - \left( \frac{E_{TH}}{E} \right)^{0.5} \right)^2 \frac{E_{TH}}{U}$$

$$Q = \frac{q_T}{1 + s_e(\epsilon)}$$

Y. Yamamura et al, *Rad.Eff.*, 11 65 (1983)  
I. Petrov et al, *Bulg.J.Phys.*, 18 203(1991)



62

62

### “Simple” Y(E) formula

for  $E < 1 \text{ keV}$ ,  $M_i > 15 \text{ amu}$

$$Y(E) = \frac{1.8}{U} \alpha(M_t / M_i) \frac{(Z_i Z_t)^{0.5}}{(Z_i^{2/3} + Z_t^{2/3})^{0.5}} \frac{(M_i M_t)^{0.5}}{(1 + M_t / M_i)^{0.5}} E^{0.5} \left( 1 - \left( \frac{E_{TH}}{E} \right)^{0.5} \right)^2$$

Eqn. A

$$Y(E) = \frac{3}{4\pi^2} \alpha(M_t / M_i) \frac{\gamma E}{U}$$

$$\gamma = \frac{4M_t M_i}{(M_t + M_i)^2}$$

Petrov, unpublished

Note: Works for pure metal targets, including self-sputtering

$$Y_{\text{selfsp}}(E) = (0.9/U) Z^{2/3} M E^{1/2} (1 - (8.5U/E)^{1/2})^2$$

63

### Energy efficiency of sputtering

Sputtering

Power of the ion beam -  $P_{\text{ion flux}} = J * E$   
 Power used for sputtering -  $P_{\text{sp}} = U * J * Y(E)$   
 Sputtering efficiency  $\eta = P_{\text{sp}} / P_{\text{ion flux}}$   
 $\eta = U * Y(E) / E$

Yamamura formula:  $\Rightarrow$   
 $\eta_{\text{max}}$  at  $E = 7 * E_{\text{th}}$   
 For  $3E_{\text{TH}} \leq E \leq 10E_{\text{TH}}$ ,  $\eta \geq 0.8\eta_{\text{max}}$

**Typical example:**  
 $U = 4 \text{ eV}$ ,  $E_{\text{TH}} = 30 \text{ eV}$ ,  $\eta_{\text{max}}$  at  $E = 210 \text{ eV}$   
 $0.8 \eta_{\text{max}}$   $E = 90 - 300 \text{ eV}$

Magnetrons work at or slightly above the optimum energy

I. Petrov et al, Bulg.J.Phys., 18 2013(1991)

64

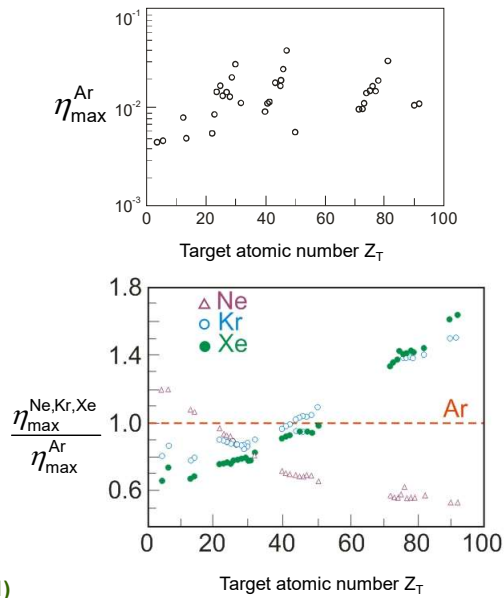
## Energy efficiency of sputtering

Sputtering

The maximum sputtering efficiency between 0.5 and 5 %

Ar provides high sputtering efficiency for a large number of metal targets (from Al to La)

Ne has ~ 20% advantage for Be and C  
Kr ~ 40-60% advantage for targets heavier than Ta



I. Petrov et al, Bulg.J.Phys., 18 2013(1991)

65

65

## Energy of Sputtered Particles

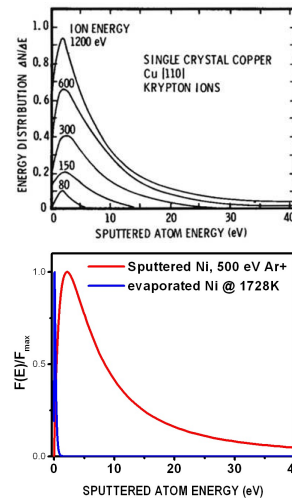
Sputtering

Sputtered atom energy has a maximum at  $\sim U/2$  (several eV) and tail extending to tens and hundreds of eV, depending on the ion energy.

Energy (Sigmund-Thompson) distribution:

$$F(E) \propto \frac{E}{(E+U)^3} \left( 1 - \left( \frac{E+U}{\gamma E_{ion}} \right)^{1/2} \right) \approx \frac{E}{(E+U)^3}$$

Has a maximum  $\sim U/2$



Comparison with thermal evaporation energies

66

66

## Energetic particles in sputterdepostion Thermalization

67

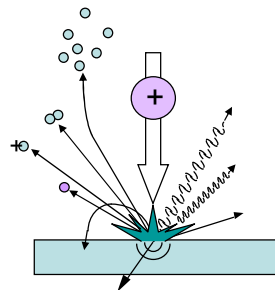
67

## Sputtering Yield: Other Species

Sputtering

Distribution of types of sputtered species:  
[Example for Ar sputtering of Cu]

Single atoms sputtered	100
Diatoms	1
Resputtered trapped gas	5
Single ions	0.1
Diatomic ions	0.001
Reflected incident species	3
Secondary electrons	10



Prof. Angus Rocket

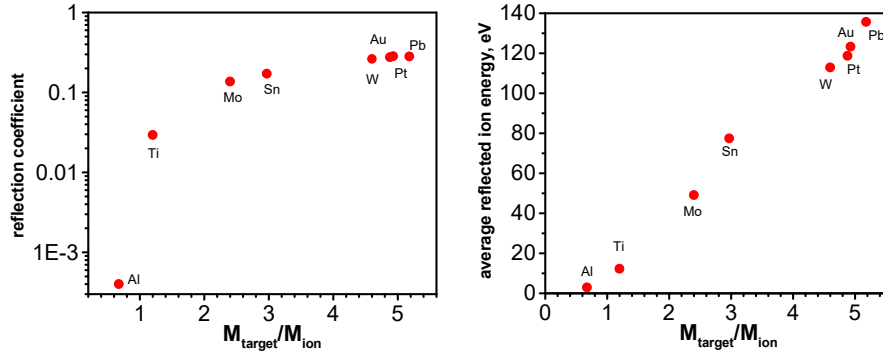
68

68

## Source of energetic particles bombardment Reflection of Primary Ions

Incident ions may be reflected from the target surface.  
Reflection coefficient = #reflected ions/#incident ions

Case study: TRIM simulation of 500 eV Ar ion scattering



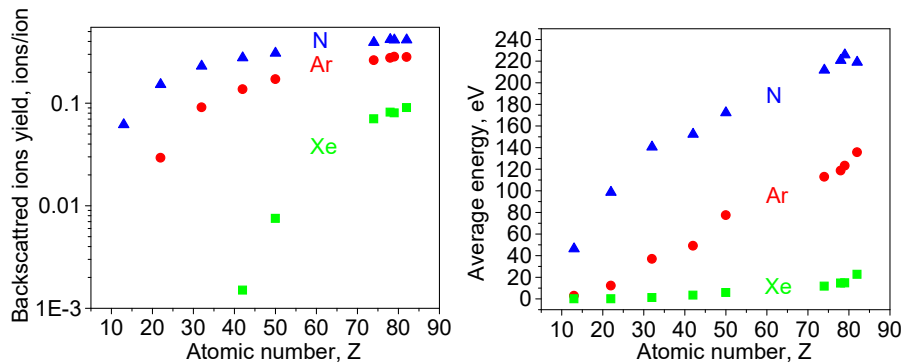
Both reflection coefficient and the average energy of the reflected ions increase when the target atom is heavier than the ion

Petrov, unpublished

## Reflection of Primary Ions, cont.

Incident ions may be reflected from the target surface.  
Reflection coefficient = #reflected ion/#incident ion

Case study: TRIM simulation of 500 eV , Xe, Ar, and N ion scattering

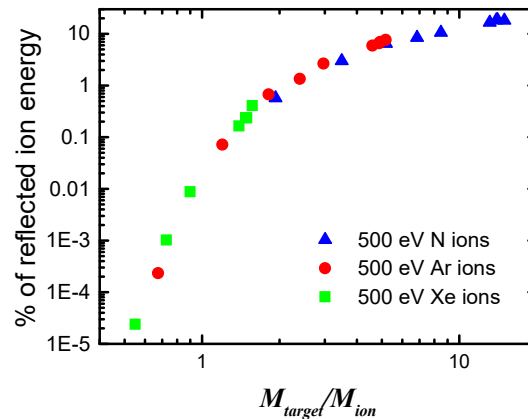


Both reflection coefficient and the average energy of the reflected ions for a given target decrease when heavier ions are used.

Petrov, unpublished

## Reflection of Primary Ions, cont.

Case study: TRIM simulation of 500 eV , Xe, Ar, and N ion scattering



A significant fraction of the incident ion energy (> 10%) is reflected back when ions are much lighter than the target atoms

Petrov, unpublished

71

71

## Another source of energetic particle bombardment The role of negative ions during sputter-deposition of oxides

**Note: Oxygen has an electron affinity of 1.461 eV**

Negative ion effects during magnetron and ion beam sputtering of  $\text{YBa}_2\text{Cu}_3\text{O}_x$

S. M. Rossnagel, and J. J. Cuomo

Citation: *AIP Conference Proceedings* **165**, 106 (1988); doi: 10.1063/1.37097

View online: <https://doi.org/10.1063/1.37097>

### ABSTRACT

The sputtering of  $\text{YBa}_2\text{Cu}_3\text{O}_x$  in both RF diode and magnetron systems has been plagued by negative ion effects. Negative ions are produced at the cathode surface during sputtering and are accelerated across the plasma sheath into the plasma. The dominant negative ion produced is  $\text{O}^-$ . The negative ion attains energy by crossing the sheath and may bombard the depositing film at a high rate. The attached electron is usually stripped in the plasma and the energetic, now neutral atom travels through the plasma and bombards the substrate location in front of the cathode. The effect of this bombardment is to reduce the net deposition rate and to alter the film composition. The negative ion yields

72

72



## Another source of energetic particle bombardment

### The role of negative ions during sputter-deposition of oxides

JAPANESE JOURNAL OF APPLIED PHYSICS  
VOL. 20, No. 3, MARCH, 1981 pp. 519-526

#### High-Energy Neutral Atoms in the Sputtering of ZnO

Kikuo TOMINAGA, Nozomu UESHIBA, Yoshihiro SHINTANI and Osamu TADA

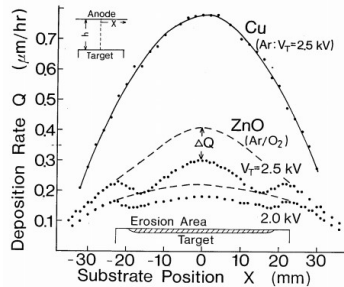


Fig. 1. Distributions of deposition rates at 0.05 Torr in DC diode sputtering, where  $V_T$  and  $I_T$  represent discharge voltage and drop in deposition rate, respectively.

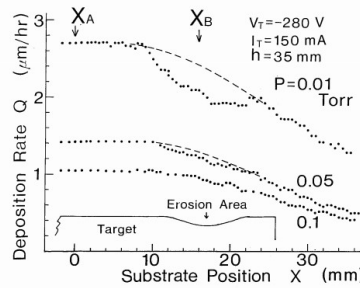
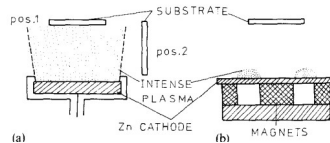


Fig. 2. Distributions of deposition rates at several gas pressures in planar magnetron sputtering.

## The role of negative ions during sputter-deposition of oxides



Placing the substrates to the side of the target and increasing the deposition pressure eliminate the adverse effect of energetic oxygen bombardment

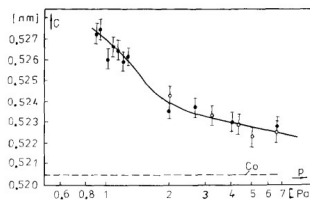


Fig. 6. Variation in the lattice parameter  $c$  of ZnO films magnetron sputtered at different total pressures and for constant  $Q = 80\%$ : ●,  $P = 55$  W; ○,  $P = 110$  W.

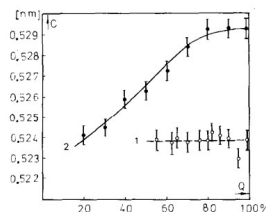


Fig. 7. Variation in the lattice parameter  $c$  of ZnO films with  $Q$  at two different total pressures and discharge powers: ●,  $p = 0.4$  Pa,  $P = 20$  W; ○,  $p = 6.6$  Pa,  $P = 110$  W.

## Transport of sputtered species in the gas phase

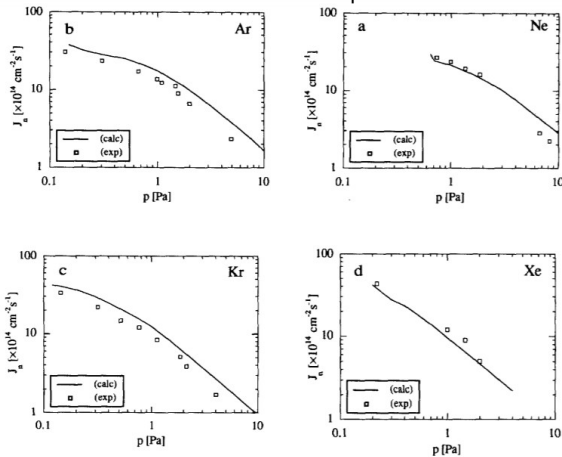
I. Petrov, I. Ivanov, V. Orlinov, and J.E. Sundgren, *J. Vac. Sci. Technol.*, 11 2733 (1993)

75

75

## Transport in the gas phase

Measured and calculated deposition fluxes



Vanadium target, 50 mm dia  
 $D_{\text{target-substrate}} = 10 \text{ cm}$

$I_T = 0.3 \text{ A}$

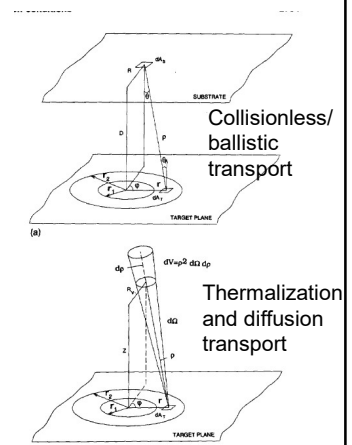
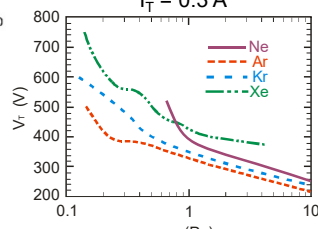


FIG. 1. Geometry used in the calculations for the ballistics transport (a), and thermalization in the gas phase (b).

Calculated curves follow pressure dependence for all gases, **only after accounting for diffusion**

### Comparison of magnetron sputter deposition conditions in neon, argon, krypton, and xenon discharges

I. Petrov,<sup>a)</sup> I. Ivanov,<sup>b)</sup> and V. Orlinov *J. Vac. Sci. Technol.*, 11 2733 (1993)  
*Institute of Electronics, Bulgarian Academy of Sciences, Sofia 1784, Bulgaria*

J.-E. Sundgren  
*Thin Film Division, Department of Physics, Linköping University, S-581 83 Linköping, Sweden*

76

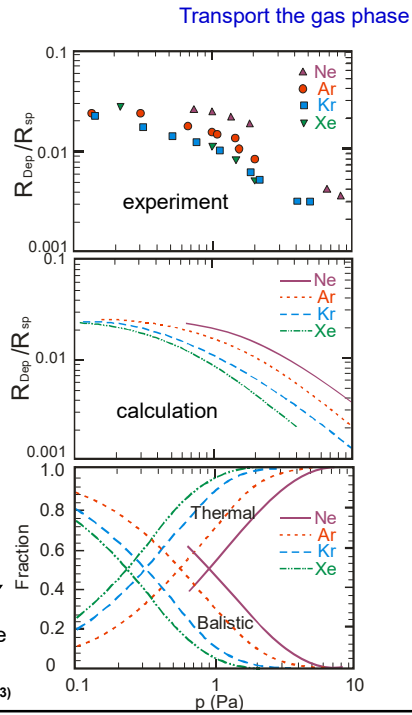
76

# Thermalization of sputtered species

In typical pressure range for magnetron sputter deposition, both collisionless and diffusive transport are effective

Note: to get the relatively good agreement at low pressures several adjustments to the mean free path had to be made  
 To get agreement at high pressures diffusive transport had to be taken into account

flux fractions in the calculation



I. Petrov, I. Ivanov, V. Orlinov, and J.E. Sundgren, J. Vac. Sci. Technol, 11 2733 (1993)

## Transport the gas phase

<http://emaps.mrl.uiuc.edu/casandra/> NIB! the site no longer works

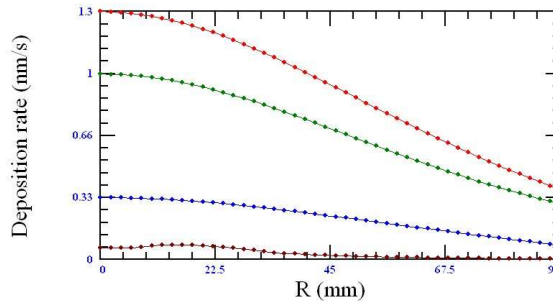


**Center for Microanalysis of Materials (CMM)**  
 A User Facility for Electron Beam Microcharacterization  
 Division of Materials Sciences/Basic Energy Sciences  
 Department of Energy

### Sputter Calculator

Copyright ©2005 I. Petrov, J. C. Mabon, Center for Microanalysis of Materials, and The Board of Trustees of the University of Illinois. All Rights Reserved.

Magnetron sputtering of Al by Ar			Legend
Gas pressure: 0.5 Pa	Inhal T: 300 K	Corrected T: 330.714 K	Total deposition rate
Discharge current: 0.5 A	Discharge voltage: 500 V		Ballistic component
Erosion disk inner radius: 10 mm	Outer radius: 30 mm	Distance to target: 100 mm	Diffusive component
			Resputtered at target



Classic paper

Calculation of deposition rates in diode sputtering systems

W. D. Westwood Journal of Vacuum Science and Technology 15, 1 (1978); <https://doi.org/10.1116/1.569429>

Mean free path

$$\lambda^{-1} = 8.34 \times 10^{14} \rho \frac{(\sigma_s + \sigma_g)^2}{4} (1 + M_s/M_g)^{1/2},$$

$\rho$  – pressure [Pa],  $\sigma_s, \sigma_g$  diameters [cm],  
 Author did not specify the temperature; it is 273 K

Number of collisions to thermalize the atom

An alternative approach is to use the concept of “persistence” introduced by Jeans.<sup>13</sup> This is the expectation that an atom of velocity  $v$  will have velocity  $v^1$  in the direction of  $v$  after the collision and is given by

$$\frac{v^1}{v} = \frac{1 - M}{1 + M} + \frac{2M}{1 + M} \frac{\ln[1 + M]^{1/2} + M^{1/2}}{4M^{3/2} (1 + M)^{1/2}} + \frac{2M^4 + 5M^3 + 3M^2 - M - 1}{4M(1 + M)^3} \quad (7)$$

The number of collisions required to reduce the initial velocity  $v_o$  to the thermal velocity  $v_g$  is

$$\eta^1 = \ln(v_g/v_o) / \ln(v^1/v) \quad (8)$$

<sup>13</sup>J. H. Jeans, in *The Dynamical Theory of Gases* (Dover New York, 1954).

Thermalization distance

$$D_3 = \eta^1 \lambda$$

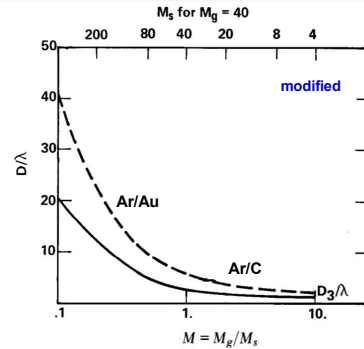


FIG. 4. The number of average collisions required to reduce a sputtered atom of mass  $M_s$  with initial energy of 5 (full lines) and 1000 eV (dashed lines) to thermal energies by collision with gas atoms, mass  $M_g$ .

The thermalization of energetic atoms during the sputtering process

Cite as: Journal of Vacuum Science & Technology A 2, 1285 (1984); <https://doi.org/10.1116/1.572396>  
 Submitted: 11 July 1983 . Accepted: 19 February 1984 . Published Online: 04 June 1998

R. E. Somekh

Department of Metallurgy and Materials Science, University of Cambridge, Pembroke Street, Cambridge, CB2 3QZ England

(Received 11 July 1983; accepted 19 February 1984)

We consider the sputter deposition process and present a calculation of the rate of energy loss of sputtered atoms and reflected neutrals due to elastic collisions with the sputtering gas atoms. Compared to previous calculations we have explicitly taken into account the strong energy dependence of the scattering cross section recently reported by Robinson. The results suggest a much less efficient energy loss rate than previously envisaged using the classical thermal cross sections. Such thermal cross sections have been used in previous calculations to explain the high sputtering pressure required for preparing high critical transition temperature superconductors Nb<sub>3</sub>Ge and Nb<sub>3</sub>Sn.



## Thermalization of sputtered species

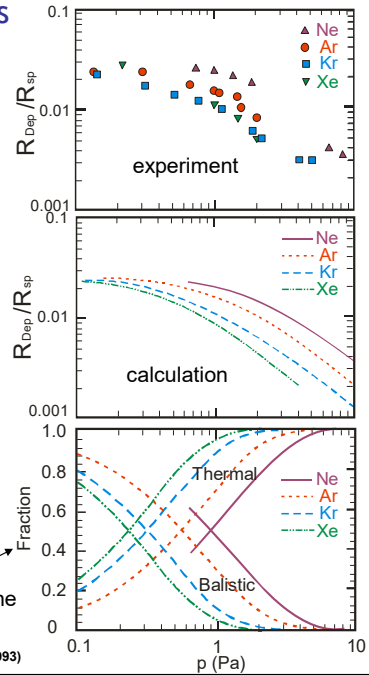
Good agreement obtained using both Westwood and Somekh approaches x10 over thermal cross-section

**In typical pressure range for magnetron sputter deposition, both collisionless and diffusive transport are effective**

**We can use pressure to tune the average energy of the sputtered species and to control the energy of the reflected ions**

flux fractions in the calculation

Transport the gas phase



I. Petrov, I. Ivanov, V. Orlov, and J.E. Sundgren, J. Vac. Sci. Technol, 11 2733 (1993)

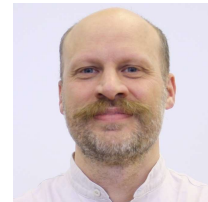
83

83



Contents lists available at SciVerse ScienceDirect

Thin Solid Films  
Thin Solid Films 520, 6337–6354 (2012)  
journal homepage: [www.elsevier.com/locate/tsf](http://www.elsevier.com/locate/tsf)



**Diederik Depla**  
Ghent University, Belgium

Special feature

Magnetron sputter deposition as visualized by Monte Carlo modeling

D. Depla\*, W.P. Leroy

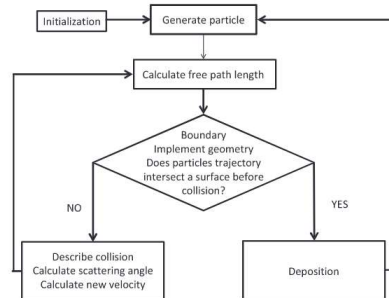
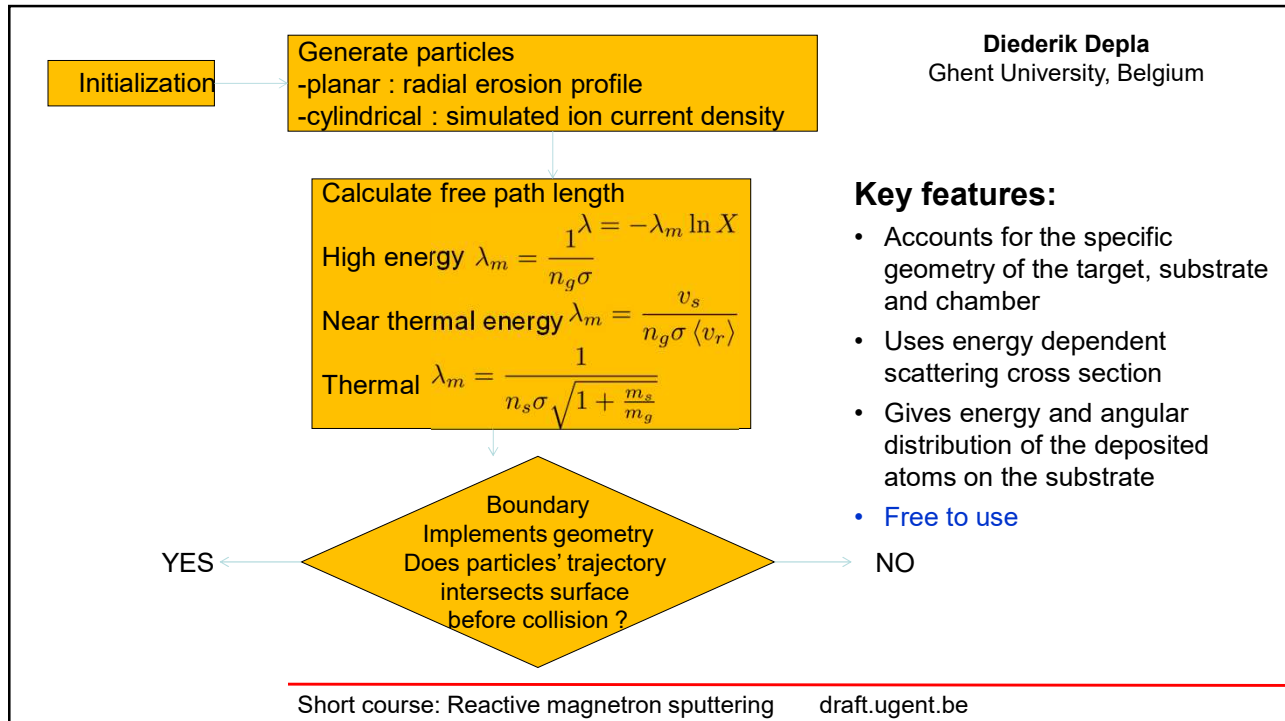


Fig. 1. Flow chart of the Monte Carlo model SIMTRA, showing the important steps in the calculation of a deposition profile.

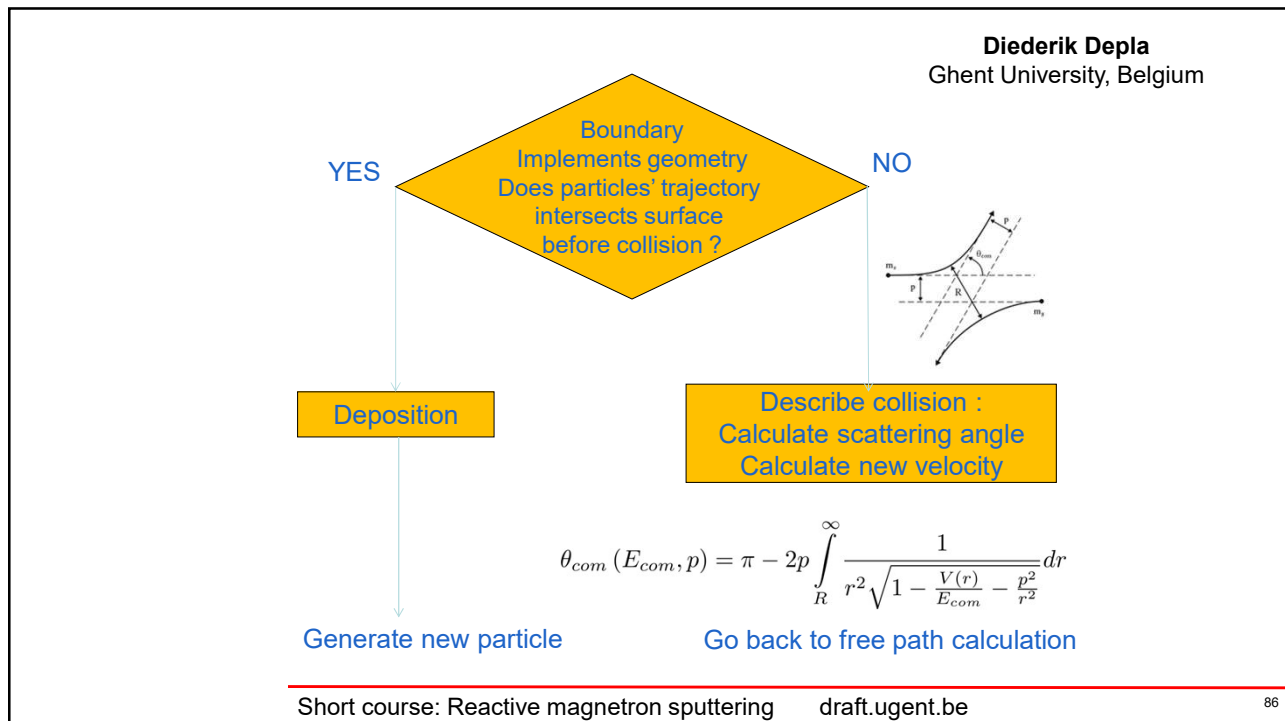
<https://www.ugent.be/we/solidstatesciences/draft/en/services/software#SiMTRA>

watch the YouTube video

84



85



86

## Influence of the sputtering gas on the deposition rate: SiMTRA comparison with literature data

D. Depla  
 Department of Solid State Sciences, Ghent University, Krijgslaan 281 (S1), 9000 Gent, Belgium

(Dated: April 26, 2023)

unpublished

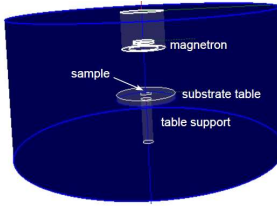
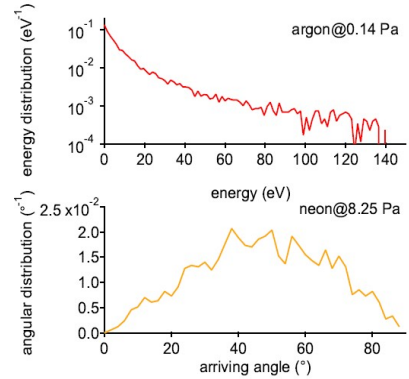
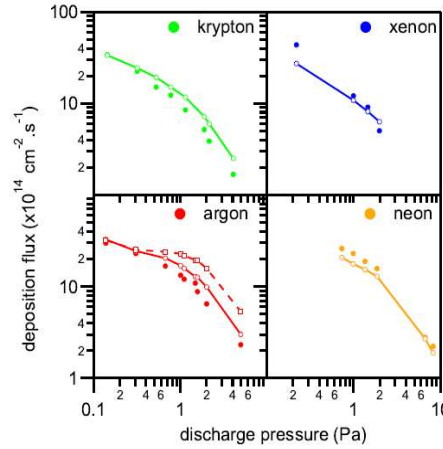
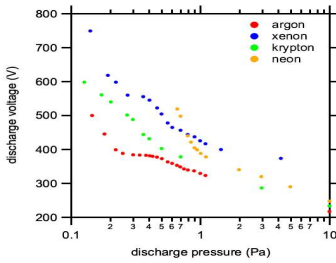


Figure 1. Used geometry. The vacuum chamber (blue) is a cylinder with a radius and a height of 0.25 m.



Exp. Data from: I. Petrov, I. Ivanov, V. Orlinov, and J.E. Sundgren, J. Vac. Sci. Technol, 11 2733 (1993)

87

Variations of sputtering systems  
 Magnetron sputtering  
 Unbalanced magnetrons

88

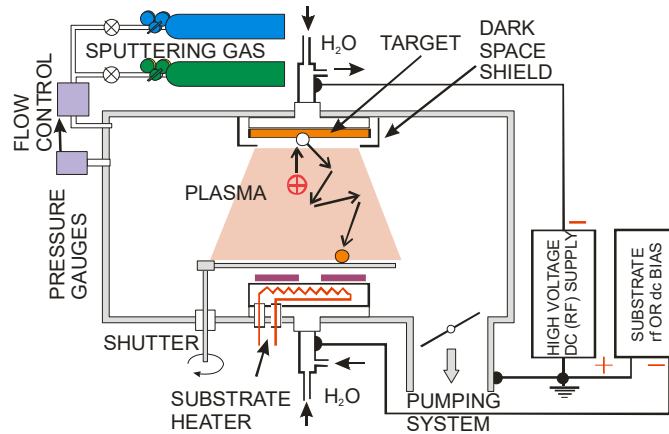
88



## Diode sputter deposition system

Components and typical parameters

$V_T \sim 2-5 \text{ kV}$   
 $J_T \sim 1 \text{ mA/cm}^2$   
 $p \sim 50-80 \text{ mTorr}$   
 $\lambda \ll d_{TS}$



89

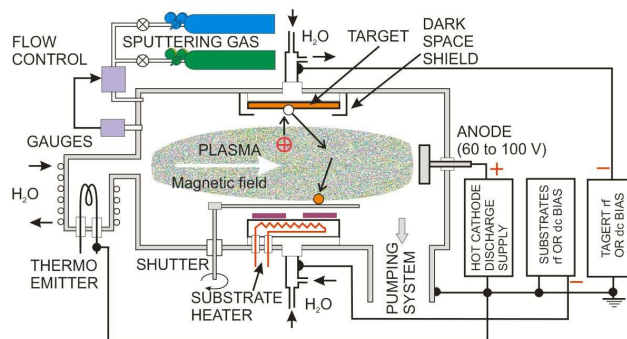
89

## Triode sputter deposition system

Components and typical parameters

$V_T \sim 1 \text{ kV}$   
 $J_T \sim 5 \text{ mA/cm}^2$   
 $p \sim 10-20 \text{ mTorr}$   
 $\lambda \sim d_{TS}$

Thermionic arc;  $V_{arc} \sim 10-30 \text{ V}$ ,  $I_{arc} \sim \text{several Amps}$   
 Independent control of ion flux and energy  
 The presence of a hot filament hampers reactive deposition



90

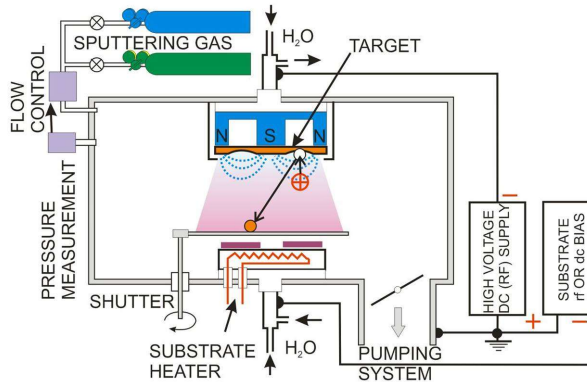
90

# Magnetron sputter deposition system

## Components and typical parameters

$V_T \sim 0.3-0.5 \text{ kV}$   
 $J_T \sim 10-100 \text{ mA/cm}^2$   
 $P_T \sim 3-50 \text{ W/cm}^2$   
 $p \sim 2-20 \text{ mTorr}$   
 $\lambda > d_{TS}, \lambda < d_{TS}$

ExB field near target enhances ionization efficiency, thus reducing both  $V_T$  and  $p$



91

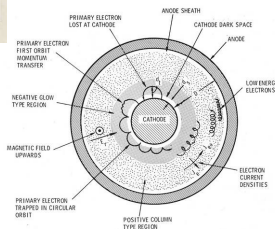
91

# Magnetron discharge characteristics

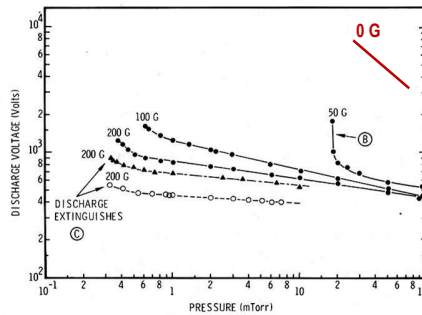
Thornton, cylindrical magnetrons



John A. Thornton  
MATSE,  
University of Illinois



Conventional DC sputtering: kV, tens mTorr

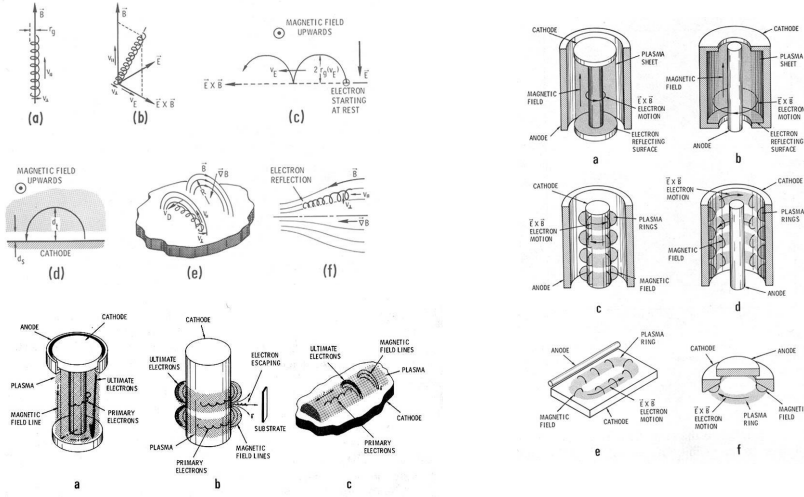


J.A. Thornton and A.S. Penfold, "Cylindrical Magnetron Sputtering," in *Thin Film Processes*, edited by J.L. Vossen and W. Kern, Academic Press, NY 1978.

92

92

# ExB configurations



$$V_T = \frac{1}{E_o \gamma_i \epsilon_i \epsilon_e}$$

$$\epsilon_i \epsilon_e \approx 1$$

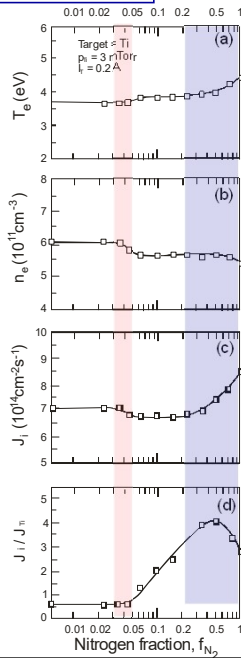
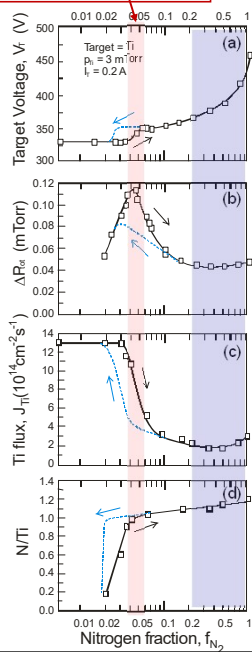
J.A. Thornton and A.S. Penfold, "Cylindrical Magnetron Sputtering," in *Thin Film Processes*, edited by J.L. Vossen and W. Kern, Academic Press, NY 1978.

## Sputtering of Ti target in Ar+N2

I. Petrov, A. Myers, J.E. Greene, and J.R. Abelson, *JVST A 12*, 2846 (1994)

Target surface reaction, transition from Ar sputtering pure metal to nitride

Transition from discharge in Ar to a discharge in N<sub>2</sub>



## Reactive sputtering

$$V_T = \frac{1}{E_o \gamma_i \epsilon_i \epsilon_e}$$

$$Y(E) = \frac{0.04}{U} a(M_t/M_i) S_H(E)$$

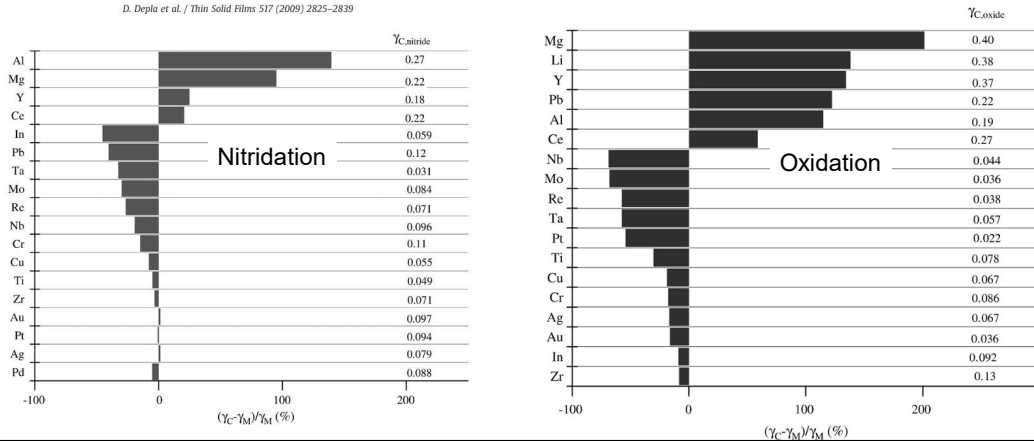
# Magnetron sputter deposition: Linking discharge voltage with target properties

D. Depla\*, S. Mahieu, R. De Gryse

Thin Solid Films 517 (2009) 2825–2839

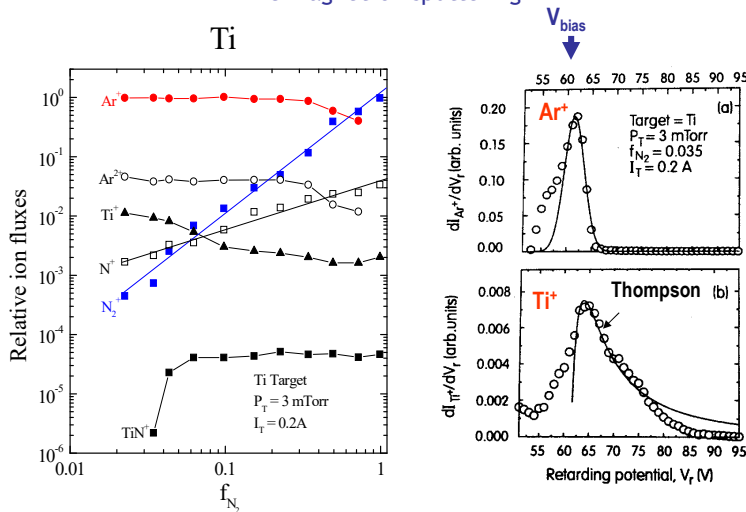
Department of Solid State Sciences, Ghent University, Krijgslaan 281(S1) 9000 Gent, Belgium

## Relative change of the effective secondary ion-electron emission coefficient upon target reaction "poisoning"



95

## Ion distribution at the substrate DC magnetron sputtering



Gas ions dominate:  
**Ar<sup>+</sup> in most of the range, N<sub>2</sub><sup>+</sup> in pure Nitrogen**

Gas ions energy corresponds to the applied bias  
 Ti<sup>+</sup> extend to higher energies due to being sputter ejected

I. Petrov, A. Myers, J.E. Greene, and J.R. Abelson, JVST A 12, 2846 (1994)

96

96

## Reactive sputtering, the big names

Modeling reactive magnetron sputtering: Opportunities and challenges

D. Depla<sup>a</sup>, K. Strijkmans, A. Dulmaa, F. Cougnon, R. Dedoncker, R. Schelfhout, I. Schramm,  
F. Moens, R. De Gryse

Department of Solid State Sciences, Ghent University, Krijgslaan 281 (S1), 9000 Gent, Belgium

**Thin Solid Films 688 (2019) 137326**

Fundamental understanding and modeling of reactive  
sputtering processes

S. Berg<sup>\*</sup>, T. Nyberg

The Angstrom Laboratory, Uppsala University, Box 534, 751 21 Uppsala, Sweden

**Thin Solid Films 476 (2005) 215–230**

Control of reactive sputtering processes

W.D. Sproul<sup>a,\*</sup>, D.J. Christie<sup>b</sup>, D.C. Carter<sup>b</sup>

<sup>a</sup>Reactive Sputtering Consulting, LLC, 3324 South Lemay Avenue, Fort Collins, CO 80525, USA  
<sup>b</sup>Advanced Energy Industries, Inc., 1625 Sharp Point Drive, Fort Collins, CO 80525, USA

**Thin Solid Films 491 (2005) 1 – 17**

**A parametric model for reactive high-power  
impulse magnetron sputtering of films**

J. Phys. D: Appl. Phys. **49** (2016) 055202 (18pp)

Tomáš Kozák<sup>1,2</sup> and Jaroslav Vičec<sup>1</sup>

97

97

## Magnetron sputter deposition High Power Pulse Magnetron Sputtering (HPPMS) High Power Impulse MS (HiPIMS)

Main new features:

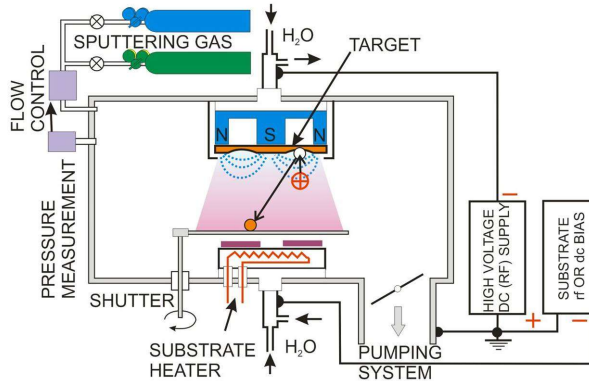
- Source of metal ions
- Time separation between the gas and the metal ions
- Higher energy ions
- Lower deposition rates

98

98

## Magnetron sputter deposition system High Power Pulse MS (HPPMS) High Power Impulse MS (HIPIMS)

**Frequency ~ 50-500 Hz**  
**Duty cycle < 10%**  
**peak values**  
 $V_T \sim 0.7-2 \text{ kV}$   
 $J_T \sim 100-1000 \text{ mA/cm}^2$   
 $P_T \sim 70-2000 \text{ W/cm}^2$   
 $p \sim 2-20 \text{ mTorr}$   
 $\lambda > d_{TS}, \lambda < d_{TS}$



Main new features:

- Source of metal ions
- Time separation between the gas and the metal ions
- Higher energy ions
- Lower deposition rates

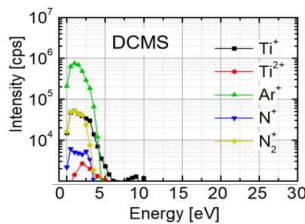
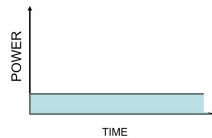
99

99

## DCMS vs HiPIMS

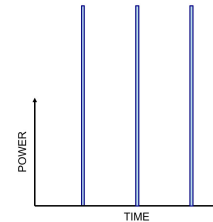
Same average power  
Same hardware

$V_T \sim 0.3-0.5 \text{ kV}$   
 $P_T \sim 10-50 \text{ W/cm}^2$   
 $p \sim 2-20 \text{ mTorr}$

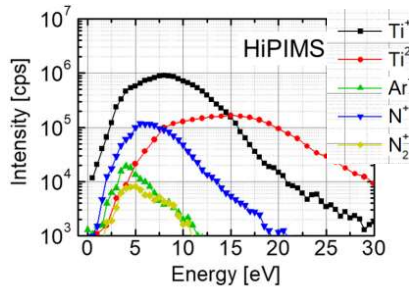


1. Thermal gas ions
2. Neutral deposited atoms
3. High deposition rate

**peak values**  
 $V_T \sim 0.7-2 \text{ kV}$   
 $P_T \sim 500-3000 \text{ W/cm}^2$   
 $p \sim 2-20 \text{ mTorr}$



**Frequency ~ 100-500 Hz**  
**Duty cycle < 1-4 %**



Hybrid HiPIMS/DCMS

1. Energetic metal ions
2. Gas and metal ions separated in time
3. Lower deposition rates

100

100

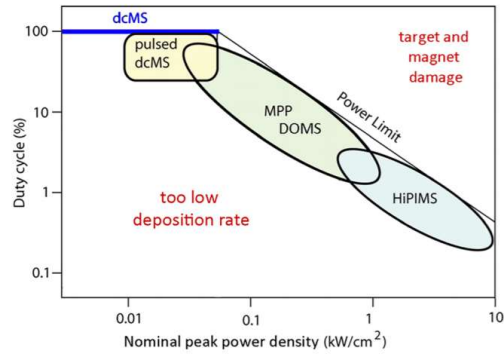


FIG. 36. Schematic showing various forms of magnetron sputtering in a duty-cycle—nominal peak power density diagram. Reprinted with permission from Gudmundsson *et al.*, *J. Vac. Sci. Technol. A* 30, 030801 (2012). Copyright 2012 AIP Publishing LLC (Fig. 1 from Ref. 140).

Gudmundsson, Brenning, Lundin, Helmersson,  
*J. Vac. Sci. Technol. A* 30 (2012) 030801

Andre Anders, Tutorial: Reactive high power impulse magnetron sputtering (R-HiPIMS) *JAP* 121, 171101 (2017)  
<https://doi.org/10.1063/1.4978350>

101

101

## High Power Impulse Magnetron Sputtering

Surface and Coatings Technology 122 (1999) 290–293

A novel pulsed magnetron sputter technique utilizing very high target power densities

Vladimir Kouznetsov <sup>a,\*</sup>, Karol Macák <sup>a</sup>, Jochen M. Schneider <sup>a</sup>, Ulf Helmersson <sup>a</sup>,  
 Ivan Petrov <sup>b</sup>



Using a novel pulsed power supply in combination with a standard circular flat magnetron source, operated with a Cu target, a peak power density of 2800 W cm<sup>-2</sup> was achieved. This results in a very intense plasma with peak ion current densities of up to 3.4 A cm<sup>-2</sup> at the substrate situated 10 cm from the target. The ionized fraction of the deposited Cu flux was estimated to be approximately 70% from deposition rate measurements. The potential for high-aspect-ratio trench filling applications by high

Source of metal ions

Both May 1999

Vacuum 53 (1999) 133–136

Impulse irradiation plasma technology for film deposition

I.K. Fetisov, A.A. Filippov, G.V. Khodachenko, D.V. Mozgrin, A.A. Pisarev\*

Department of Plasma Physics, Moscow State Engineering and Physics Institute, Kashirskoe shosse, 115409 Moscow, Russia

A new type of the sputtering discharge and its applications for film deposition are described in the paper. This is a pulse, quasi-stationary and high-current diffuse discharge in the magnetic field. The combination of its properties makes the discharge to be a promising instrument for film deposition. Complex films of high quality have been obtained by using this discharge. Features and

102



# High Power Impulse Magnetron Sputtering

Surface and Coatings Technology 122 (1999) 290–293

A novel pulsed magnetron sputter technique utilizing very high target power densities

Vladimir Kouznetsov <sup>a,\*</sup>, Karol Macák <sup>a</sup>, Jochen M. Schneider <sup>a</sup>, Ulf Helmersson <sup>a</sup>, Ivan Petrov <sup>b</sup>



Source of metal ions

Using a novel pulsed power supply in combination with a standard circular flat magnetron source, operated with a Cu target, a peak power density of  $2800 \text{ W cm}^{-2}$  was achieved. This results in a very intense plasma with peak ion current densities of up to  $3.4 \text{ A cm}^{-2}$  at the substrate situated 10 cm from the target. The ionized fraction of the deposited Cu flux was estimated to be approximately 70% from deposition rate measurements. The potential for high-aspect-ratio trench filling applications by high



Both May 1999



Vacuum 53 (1999) 133–136

Impulse irradiation plasma technology for film deposition

I.K. Fetisov, A.A. Filippov, G.V. Khodachenko, D.V. Mozgrin, A.A. Pisarev\*

Department of Plasma Physics, Moscow State Engineering and Physics Institute, Kashirskoe shosse, 115409 Moscow, Russia

A new type of the sputtering discharge and its applications for film deposition are described in the paper. This is a pulse, quasi-stationary and high-current diffuse discharge in the magnetic field. The combination of its properties makes the discharge to be a promising instrument for film deposition. Complex films of high quality have been obtained by using this discharge. Features and

103

Surface & Coatings Technology 293 (2016) 42–47

## High-current impulse magnetron discharge with liquid target

Alexander V. Tumarkin <sup>\*</sup>, Andrey V. Kaziev, Maxim M. Kharkov, Dobrynya V. Kolodko, Igor V. Ilychev, Georgy V. Khodachenko <sup>1</sup>

Department of Plasma Physics, National Research Nuclear University MEPhI (Moscow Engineering Physics Institute), 31 Kashirskoye Shosse, Moscow 115409, Russia

### 1. Introduction

The HiPIMS technology has been extensively developed over the recent two decades, and nowadays it is widely used for the deposition of superior thin films. The coatings prepared by HiPIMS find their applications in advanced areas of industry [1]. The HiPIMS technique emerged in 1999 [2] and quickly attracted major attention both among the industries and the academia. However, the pioneering studies in the field of the high power pulsed magnetron sputtering were carried out in the beginning of 1990s [3–5]. In particular, Mozgrin et al. [4] originally described the high-current impulse magnetron sputtering discharge (HCIMD), whose distinguishing feature is the behaviour of the voltage and current waveforms exhibiting the quasi-stationary stage duration up to tens ms (see Fig. 1). The pulsed discharge regimes are superior to di-

- [2] V. Kouznetsov, K. Macák, J.M. Schneider, U. Helmersson, I. Petrov, A novel pulsed magnetron sputter technique utilizing very high target power densities, Surf. Coat. Technol. 122 (1999) 290–293.
- [3] I.K. Fetisov, G.V. Khodachenko, D.V. Mozgrin, Quasi-stationary High Current Forms of Low Pressure Discharge in Magnetic Field, Proc. ICPG-XX, Piza, Italy, 1991 476–478.
- [4] D.V. Mozgrin, I.K. Fetisov, G.V. Khodachenko, High-current low-pressure quasi-stationary discharge in a magnetic field: experimental research, Plasma Phys. Rep. 21 (1995) 422–433.
- [5] S.P. Bugaev, N.N. Koval, N.S. Sochugov, A.N. Zakharov, Investigation of a high-current pulsed magnetron discharge initiated in the low-pressure diffuse arc plasma, Proc. XVII Int. Symp. on Disch. and Electr. Insulation in Vac., Berkeley, USA 1996, pp. 1074–1076.

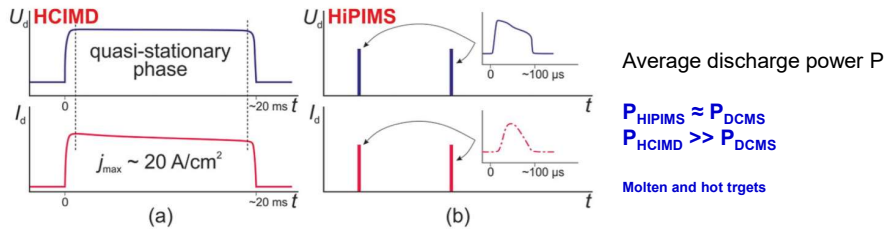


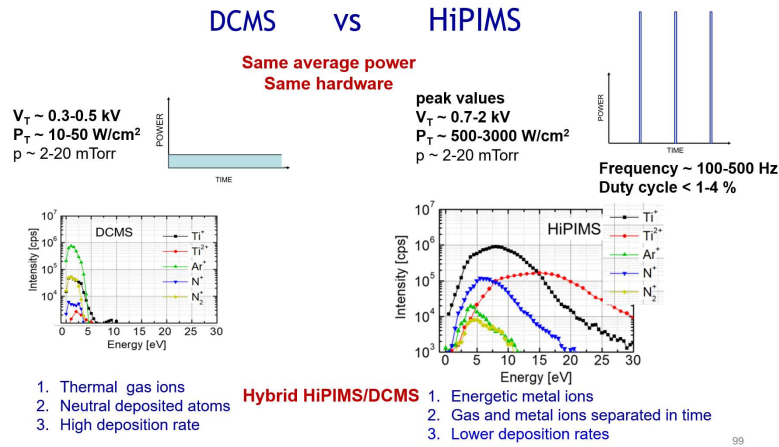
Fig. 1. Typical voltage and current traces of HCIMD (a) and HiPIMS (b).

petrov@illinois.edu

104



To properly exploit the advantages of HiPIMS it is necessary to know the energy and time evolution of the gas and metal ions



105

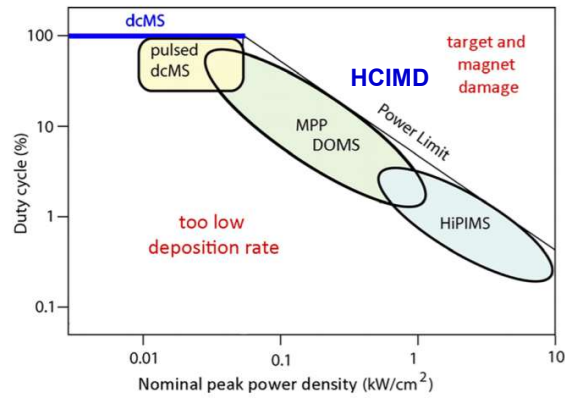


FIG. 36. Schematic showing various forms of magnetron sputtering in a duty-cycle—nominal peak power density diagram. Reprinted with permission from Gudmundsson *et al.*, *J. Vac. Sci. Technol. A* **30**, 030801 (2012). Copyright 2012 AIP Publishing LLC (Fig. 1 from Ref. 140).

Gudmundsson, Brenning, Lundin, Helmersson,  
*J. Vac. Sci. Technol. A* **30** (2012) 030801

Andre Anders, Tutorial: Reactive high power impulse magnetron sputtering (R-HIPIMS) *JAP* **121**, 171101 (2017)  
<https://doi.org/10.1063/1.4978350>

106

106

### Modulated Pulse Power Magnetron Sputtering MPPMS

J. Lin et al. / Surface & Coatings Technology 203 (2009) 3676–3685

3677

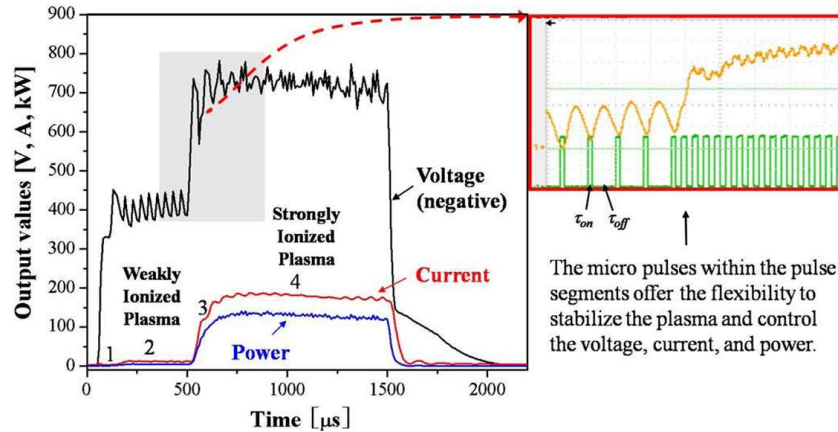
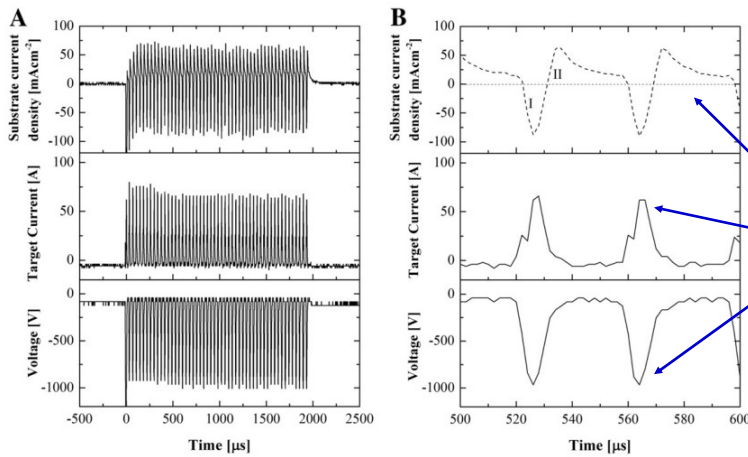


Fig. 1. Typical MPP pulse showing four steps of the target voltage, current and power evolutions during one modulated pulse (1500 μs overall pulse length in this example): 1) ignition of the weakly ionized plasma; 2) duration of the weakly ionized plasma; 3) transition stage from the weakly ionized plasma to the strongly ionized plasma; (4) duration of the strongly ionized plasma. The inserted figure on the right side shows the micro pulses within the pulse with adjustable voltage on ( $\tau_{on}$ ) and off ( $\tau_{off}$ ) times.

107

### Deep Oscillation Magnetron Sputtering DOMS

Lin et al, Surface and Coatings Technology (2015) 276, pp. 70-76



Scopus:  
A few dozen of publications  
on MPP and DOMS

Typical parameters:  
Substrate current  
Current  
Voltage  
Macropulses - ~0.5-1.5 ms, 100s Hz  
Micropulses ~20-30 kHz

“The DOMS technique offers virtually **arc-free conditions** for reactive sputtering of many **insulating films** e.g. AlN, Al<sub>2</sub>O<sub>3</sub>, Si<sub>3</sub>N<sub>4</sub>, SiO<sub>2</sub> etc.”

108

## Ionized sputter deposition using an extremely high plasma density pulsed magnetron discharge

Karol Macák, Vladimir Kouznetsov, Jochen Schneider, and Ulf Helmersson<sup>a)</sup>  
*Department of Physics, Linköping University, SE-581 83 Linköping, Sweden*

Ivan Petrov  
*Materials Science Department and Materials Research Laboratory, University of Illinois,*

1533 J. Vac. Sci. Technol. A 18(4), Jul/Aug 2000

**Transition from gas to metal vapor discharge/gas rarefaction;  
 Time separation between the gas and the metal ions**

Ar gas rarefaction > 80%  
 Metal plasma density >  $10^{12} \text{ cm}^{-3}$

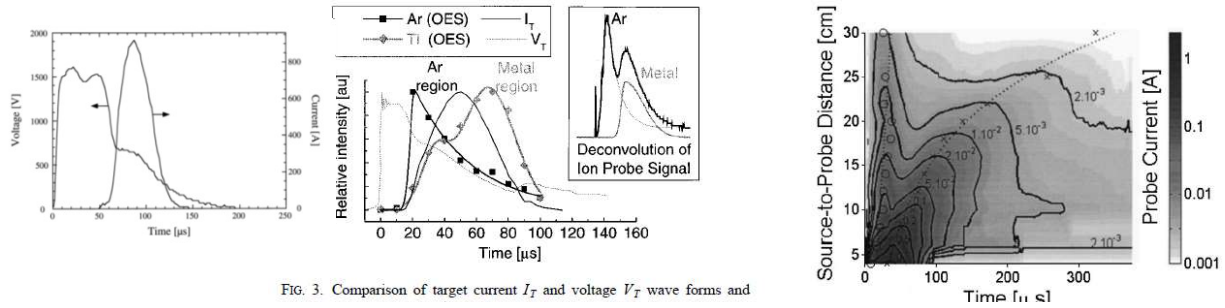


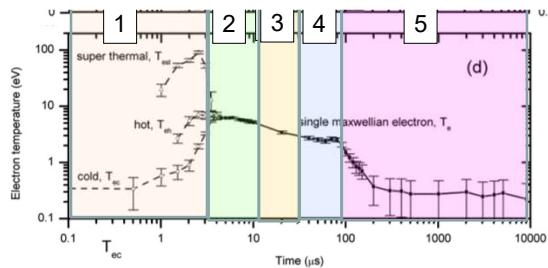
FIG. 3. Comparison of target current  $I_T$  and voltage  $V_T$  wave forms and

**The existence of time separation between the Ar and metal-ion dominated fluxes at the substrate opens the possibility for selection one of the components for ion-assisted by using a pulsed bias voltage with suitable synchronization**

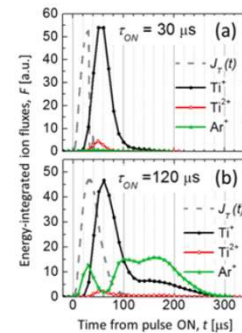
109

## Phases of the HIPIMS discharge

1. Ignition/discharge breakdown
2. Current rise
3. Gas depletion/rarefaction
4. Metal mode\*
5. Afterglow



P Poolcharuansin and J W Bradley  
 Plasma Sources Sci. Technol. 19 (2010) 025010



110

# Time- and Species-Resolved Plasma Imaging as a New Diagnostic Approach for HiPIMS Discharge Characterization

Matěj Hála, Oleg Zabeida, Bill Baloukas, Jolanta E. Klemberg-Sapieha, and Ludvík Martinu

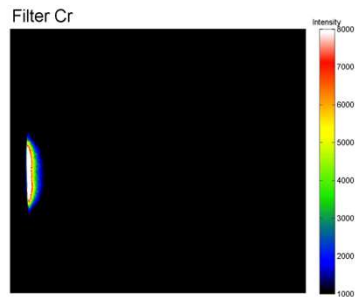
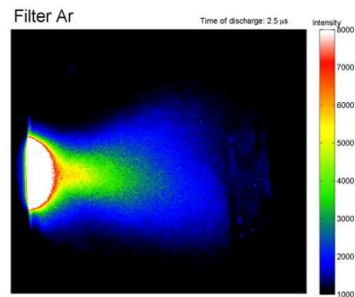
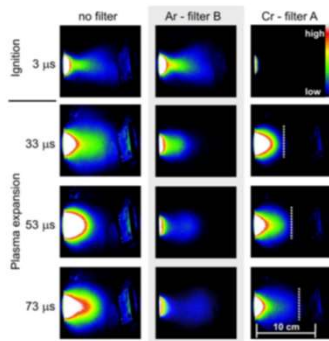
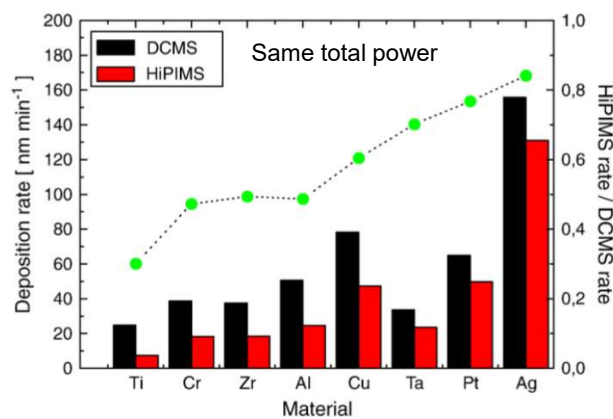


Fig. 4. Images taken during 200- $\mu$ s HiPIMS discharge pulses in pure Ar at 4 Pa with no filter on the camera objective (left column), and with an optical bandpass filter specific for emission from neutral Ar (middle column - filter B), and from neutral Cr (right column - filter A). The images were recorded at different times after the pulse initiation as indicated in Fig. 3. For clarity purposes, different colors are assigned to different emission intensities, according to the scale depicted in the upper right image. The position of the Cr emission wavefront is highlighted by a dashed line. Please note that a movie of the discharge can be downloaded at <http://ieeexplore.ieee.org>.

111

## Lower deposition rates in HiPIMS due to back attraction of positive ions of the sputtered materials back to the target



**Note: This is just typical example**

The reduction of deposition rate may vary significantly for a given material depending on the experimental set-up and the HiPIMS characteristics, as discussed later

Fig. 1. The deposition rates for DCMS and HiPIMS discharges plotted as bars for the different target materials used (left axis). The deposition rate of HiPIMS over DCMS deposition rate is shown as a scatter plot (right axis).

M. Samuelsson, D. Lundin, J. Jensen, M. A. Raadu, J. T. Gudmundsson, and U. Helmersson, *Surf. Coat. Technol.* 202, 591 (2010).

112

## Gas sputtering and recycling, metal self-sputtering

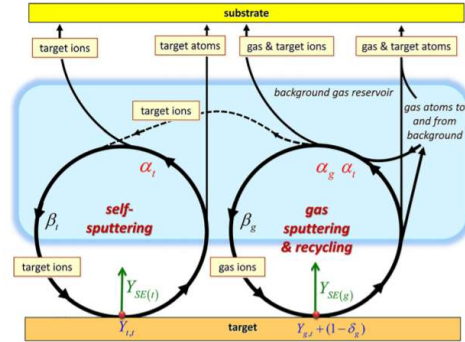
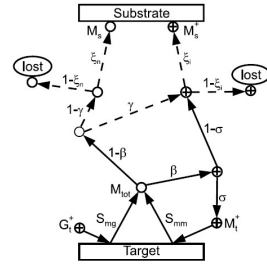
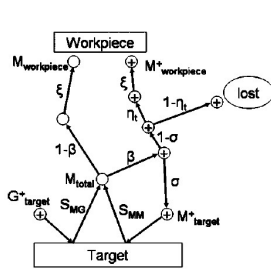


FIG. 19. Illustration of fluxes including self-sputtering, sputtering by gas, and repeated use ("recycling") of gas atoms. Figures like that illustrate a system of nonlinear rate equations. Reprinted with permission from Anders *et al.*, *J. Phys. D: Appl. Phys.* **45**, 012003 (2012). Copyright 2012 the IOP Publishing Ltd. (Fig. 1 from Ref. 84).

DJ Christie, *JVST A* 23,330, 2005

J Vlcek K Burkalova  
*Plasma Sources Sci Technol* 19, 065010, 2010

A Anders, J Capek, M Hala and L Martinu  
*J Phys D* 45, 012003 (2012)

113

## A unified treatment of self-sputtering, process gas recycling, and runaway for high power impulse sputtering magnetrons

N Brenning<sup>1,2,3</sup>, J T Gudmundsson<sup>1,2,4</sup>, M A Raadu<sup>1</sup>, T J Petty<sup>2</sup>, T Minea<sup>2</sup> and D Lundin<sup>2,5</sup>

*Plasma Sources Sci. Technol.* **26** (2017) 125003 (13pp)

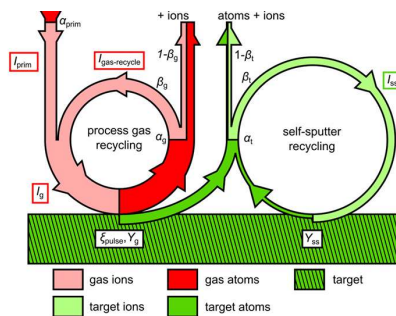


Figure 1. A schematic illustration, revised from Anders *et al.* [9], of the combined processes of process gas recycling and self-sputter recycling. The widths of the flow arrows are drawn to scale with a parameter combination  $\eta_{\text{gas}} = 1$ ,  $\zeta_{\text{gas}} = 1$ ,  $\alpha_0 = 0.7$ ,  $\beta_0 = 0.7$ ,  $\gamma_0 = 0.4$ ,  $\alpha_1 = 0.8$ ,  $\beta_1 = 0.7$ , and  $\gamma_{\text{SS}} = 0.5$ . This combination is arbitrarily chosen as suitable to illustrate combined gas-recycling and SS-recycling.

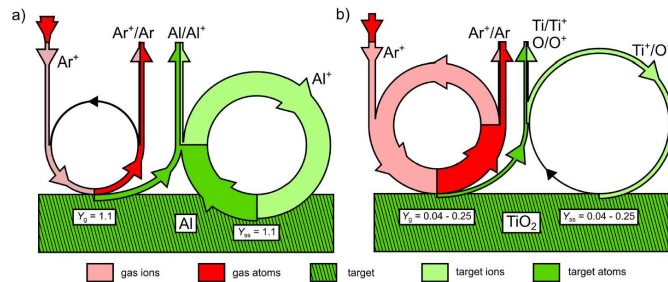


Figure 7. Recycling loops such as in figure 1, with the widths of the arrows drawn to scale for two cases from figure 6: one (a) showing the discharge with an Al target where SS-recycling dominates, and one (b) showing the discharge with a TiO<sub>2</sub> target where gas-recycling dominates.

Inert gas sputtering and self-sputtering yield important

114

114

## Time evolution and the energy distribution of the gas and metal ion fluxes in HiPIMS by in-situ ion mass spectrometry measurements

- Time separation between the gas and the metal ions (gas rarefaction)  
important for HiPIMS with metal ions synchronized bias and bipolar HiPIMS
- Control of doubly charged ions in HIPIMS  
important for metastable films synthesis (examples tomorrow)
- High energy metal ions in HIPIMS

115



Prof. Greg Greczynski

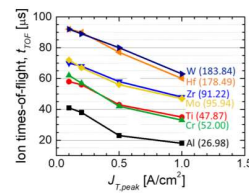
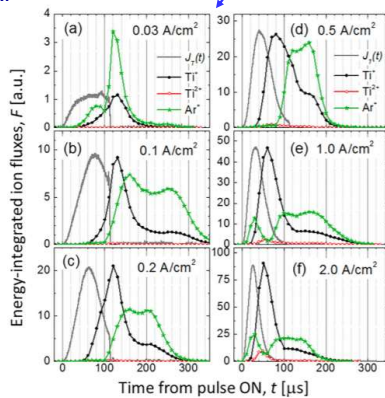
### Gas rarefaction effects during high power pulsed magnetron sputtering of groups IVb and VIb transition metals in Ar

JVST A, 35, 060601 (2017)

Grzegorz Greczynski, Igor Zhirkov, Ivan Petrov, J. E. Greene, and Johanna Rosen

#### Time separation between the gas and the metal ions

Depends on the peak current, on the material (next slide), and on the pulse length



Time of flight (average ion velocity) depends on the peak current and ion mass

The HiPIMS pulse length is 120  $\mu\text{s}$  at a frequency of 300 Hz.

116

116



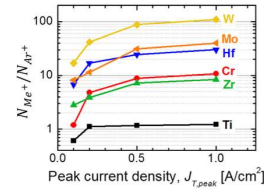
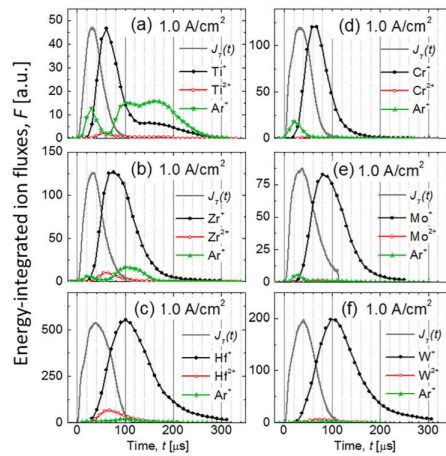
**Gas rarefaction effects during high power pulsed magnetron sputtering of groups IVb and VB transition metals in Ar**

Grzegorz Greczynski, Igor Zhirkov, Ivan Petrov, J. E. Greene, and Johanna Rosen

JVST A, 35, 060601 (2017)

**Time separation between the gas and the metal ions**  
Depends on the peak current, on the material, on the pulse length (next slide)

IV B	V B	VI B
22 <b>Ti</b> Titanium 47.867	23 <b>V</b> Vanadium 50.942	24 <b>Cr</b> Chromium 51.996
40 <b>Zr</b> Zirconium 91.224	41 <b>Nb</b> Niobium 92.906	42 <b>Mo</b> Molybdenum 95.95
72 <b>Hf</b> Hafnium 178.49	73 <b>Ta</b> Tantalum 180.95	74 <b>W</b> Tungsten 183.84



The ratio of the gas to metal ion fluxes depends primarily on ion type (mainly mass ratio and the self-sputtering yield)

117

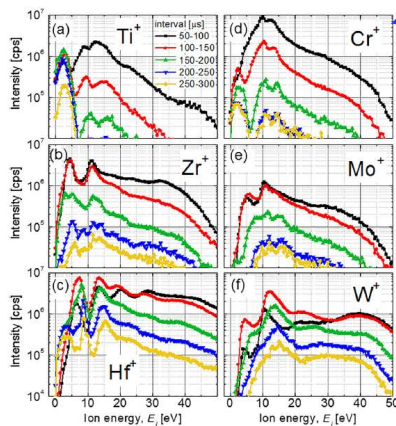
117

**Gas rarefaction effects during high power pulsed magnetron sputtering of groups IVb and VIb transition metals in Ar**

Grzegorz Greczynski, Igor Zhirkov, Ivan Petrov, J. E. Greene, and Johanna Rosen

JVST A, 35, 060601 (2017)

**Time separation between the gas and the metal ions**  
Depends on the peak current, on the material, on the pulse length (next slide)



- The ions of lighter elements, Ti and Cr lose energy due to back filling with Ar
- The ions of the heavy elements Hf and W preserve the high energy tails **more pronounced rarefaction and more difficult to thermalize**

**The heavy metals promote gas rarefaction and high metal to gas ion ratio – an advantage in metal ion assisted film growth**

FIG. 4. (Color online)  $Me^+$  ion energy distribution functions (IEDFs) recorded at the substrate position during HIPIMS sputtering of  $Me = Ti$  (a),  $Zr$  (b),  $Hf$  (c),  $Cr$  (d),  $Mo$  (e), and  $W$  (f) targets in Ar at 0.4 Pa (3 mTorr). The IEDFs are acquired during 50- $\mu s$  time intervals over the time period from 0 (pulse ignition) to 300  $\mu s$ . The pulse length is 120  $\mu s$ .

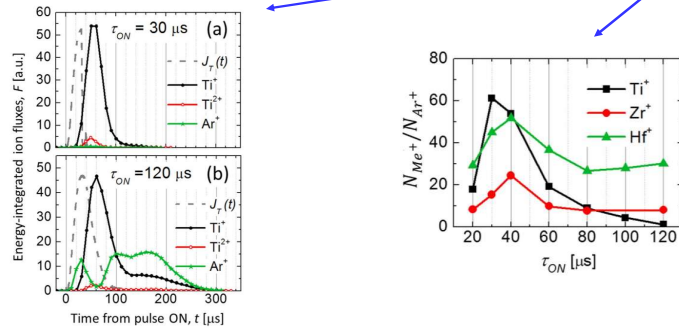
118

118

Control of the metal/gas ion ratio incident at the substrate plane during high-power impulse magnetron sputtering of transition metals in Ar

G. Greczynski<sup>a,\*,c</sup>, I. Zhirkov<sup>a</sup>, I. Petrov<sup>a,b</sup>, J.E. Greene<sup>a,b,c</sup>, J. Rosen<sup>a</sup> *Thin Solid Films* 642 (2017) 36–40

**Time separation between the gas and the metal ions**  
 Depends on the peak current, on the material, on the pulse length )



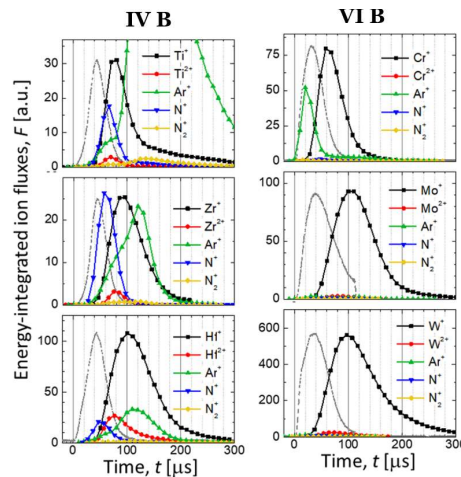
119

119

**Reactive sputtering: Ion mass spectrometry at substrate position to determine the metal ion dominated phase**

IV B	V B	VI B
22 <b>Ti</b> Titanium 47.867	23 <b>V</b> Vanadium 50.942	24 <b>Cr</b> Chromium 51.996
40 <b>Zr</b> Zirconium 91.224	41 <b>Nb</b> Niobium 92.906	42 <b>Mo</b> Molybdenum 95.95
72 <b>Hf</b> Hafnium 178.49	73 <b>Ta</b> Tantalum 180.95	74 <b>W</b> Tungsten 183.84

p = 3 mTorr; N<sub>2</sub>/Ar flow ratio of 0.11  
 Peak Current 1 A/cm<sup>2</sup>



Ar<sup>+</sup> overlaps with Me<sup>+</sup>    Ar<sup>+</sup> separated from Me<sup>+</sup>

Grzegorz Greczynski, Igor Zhirkov, Ivan Petrov, J. E. Greene, and Johanna Rosen, *JVST A* 36, 020602 (2018);

120

120

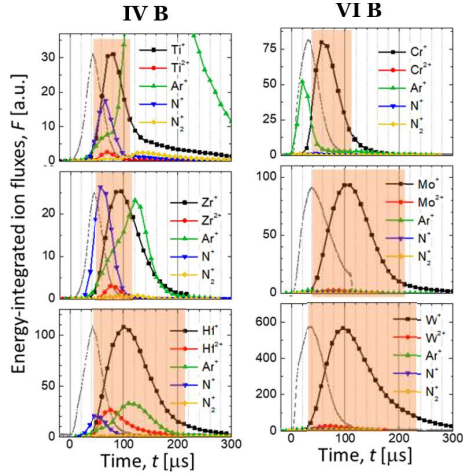


Reactive sputtering: Ion mass spectrometry at substrate position to determine the metal ion dominated phase

**Metal ion synchronized bias**

IV B	V B	VI B
22 <b>Ti</b> Titanium 47.867	23 <b>V</b> Vanadium 50.942	24 <b>Cr</b> Chromium 51.996
40 <b>Zr</b> Zirconium 91.224	41 <b>Nb</b> Niobium 92.906	42 <b>Mo</b> Molybdenum 95.95
72 <b>Hf</b> Hafnium 178.49	73 <b>Ta</b> Tantalum 180.95	74 <b>W</b> Tungsten 183.84

p=3 mTorr; N<sub>2</sub>/Ar flow ratio of 0.11  
Peak Current 1 A/cm<sup>2</sup>



Ar<sup>+</sup> overlaps with Me<sup>+</sup>    Ar<sup>+</sup> separated from Me<sup>+</sup>

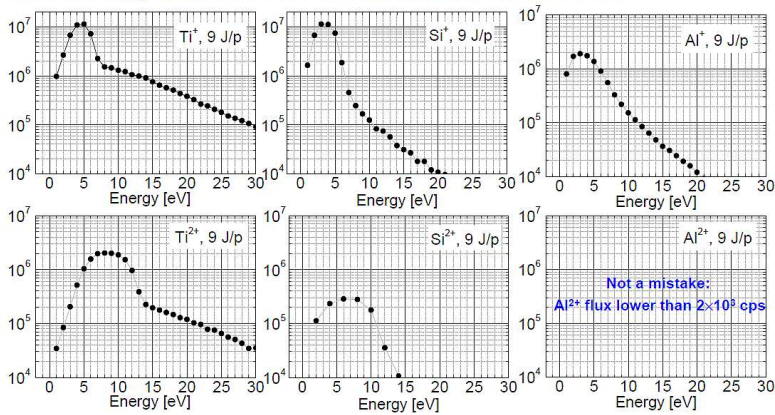
Grzegorz Greczynski, Igor Zhirkov, Ivan Petrov, J. E. Greene, and Johanna Rosen, JVST A 36, 020602 (2018);

121

121

Changes in the Me<sup>2+</sup> component  
Important for metastable materials synthesis

Ion energy distribution functions for metal ions



▪ Second ionization potential: Ti: 13.62 eV, Si: 16.35 eV, Al: 18.89 eV

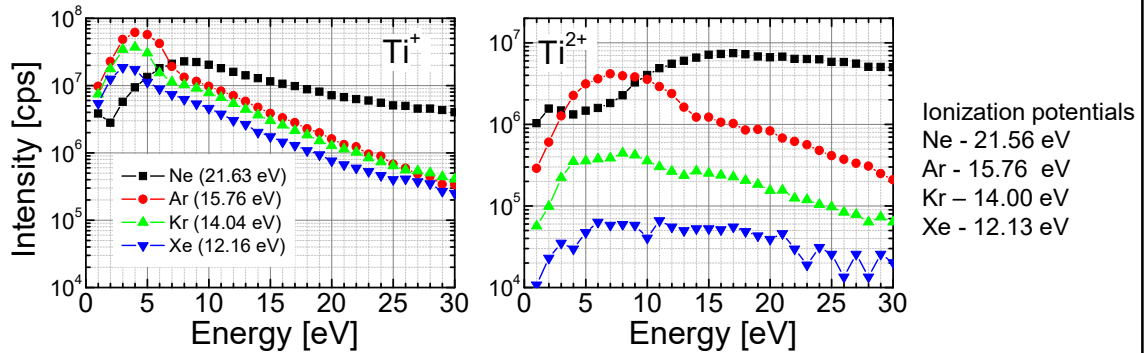


Ar first ionization potential 15.76 eV

122

## Control of doubly charged ions in HIPIMS

Model system: Ti sputtered in Ne, Ar, Kr, and Xe

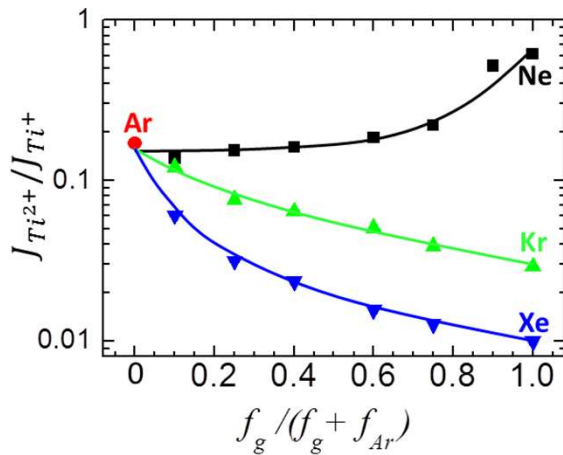


- $Ti^{2+}$  detrimental to TiAlN films grown by HIPIMS/DCMS
- Reason for high  $Ti^{2+}$  population while sputtering in Ar:  $IP_{Ti}^2 < IP_{Ar}^1$
- exchanging Ar for lower IP gases (Kr, Xe) decreases  $T_e$  and the relative amount of  $Ti^{2+}$ , while the opposite is valid for Ne which has highest IP

G. Greczynski, I. Petrov, J.E. Greene, and L. Hultman, *Vacuum* 116 (2015) 36

123

## $J_{Ti^{2+}}/J_{Ti^+}$ : comparison for Ne, Ar, Kr and Xe

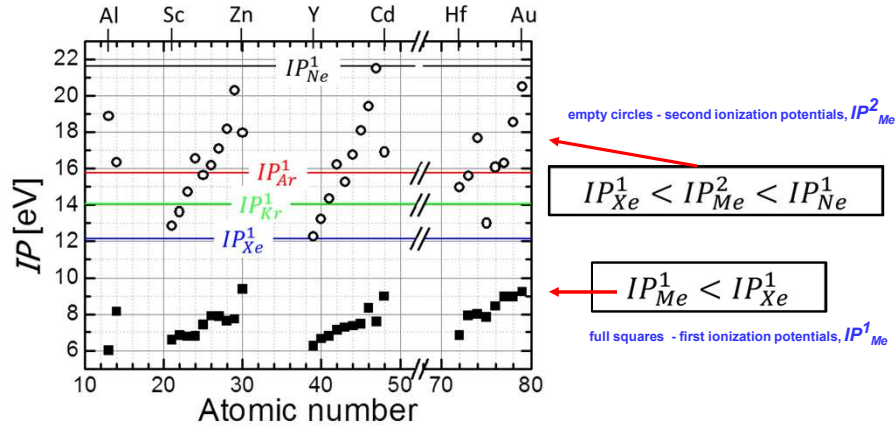


- by changing noble-gas mixtures,  $J_{Ti^{2+}}$  varies by more than two orders of magnitude with only a small change in  $J_{Ti^+}$ .
- This allows the ratio  $J_{Ti^{2+}}/J_{Ti^+}$  to be continuously tuned from 0.01 with Xe to 0.62 with Ne

G. Greczynski, I. Petrov, J.E. Greene, and L. Hultman, *Vacuum* 116 (2015) 36

124

## Extension to other materials



G. Greczynski, I. Petrov, J.E. Greene, and L. Hultman, *Vacuum* 116 (2015) 36

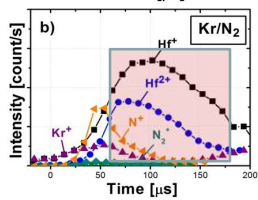
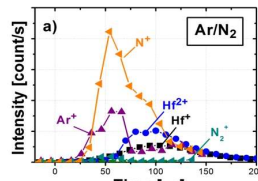
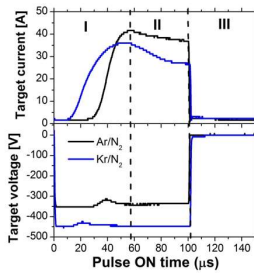
125

## An example of the importance of the choice of inert sputtering gas

Low temperature epitaxial growth of HfN/MgO(100) via HiPIMS-synchronized pulsed substrate bias

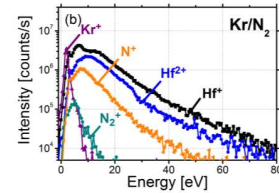
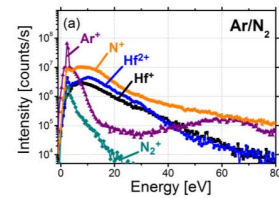
**HfN** the highest melting point  
TM nitride:  $T_M \sim 3520$  K

$T_S < 70$  °C;  $T_S/T_M < 0.1$



$V_s = -100$  V

$IP_{Ar} = 15.8$  eV  
 $IP_N = 14.5$  eV  
 $IP_{N2} = 15.6$  eV  
 $IP_{Kr} = 14.0$  eV



M.M.S. Villamayor, T. Shimizu, J. Keraudy, R.P.B. Viloan, R. Boyd, D. Lundin, J.E. Greene, Ivan Petrov, Ulf Helmersson *J. Vac. Sci. Technol. A*, Vol. 36 (2018) 06151

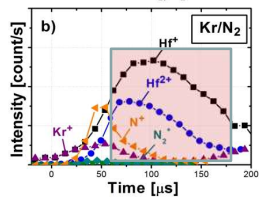
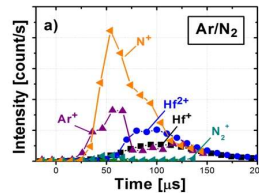
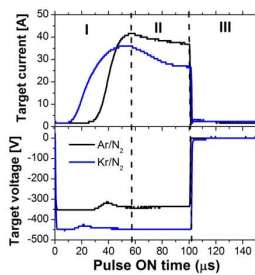
126

## An example of the importance of the choice of inert sputtering gas

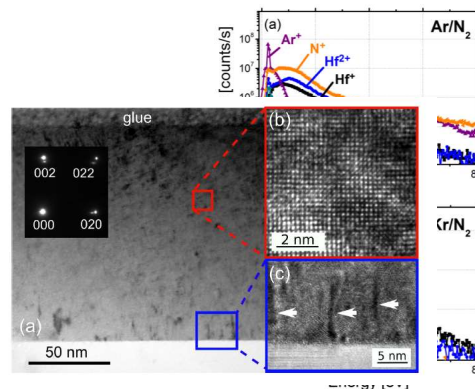
### Low temperature epitaxial growth of HfN/MgO(100) via HiPIMS-synchronized pulsed substrate bias

**HfN the highest melting point  
TM nitride:  $T_M \sim 3520$  K**

**$T_S < 70$  °C;  $T_S/T_M < 0.1$**



$V_s = -100$  V



M.M.S. Villamayor, T. Shimizu, J. Keraudy, R.P.B. Viloan, R. Boyd, D. Lundin, J.E. Greene, Ivan Petrov, Ulf Helmersson *J. Vac. Sci. Technol. A*, Vol. 36 (2018) 06151

127

## HiPIMS: source of energetic metal ions

J. Bohlmark, M. Lattemann, J.T. Gudmundsson, A.P. Ehasarian,  
Y. Aranda Gonzalvo, N. Brenning, U. Helmersson  
The ion energy distributions and ion flux composition from a high-power impulse  
magnetron sputtering discharge  
*Thin Solid Films* 515 (2006) 1522-1526

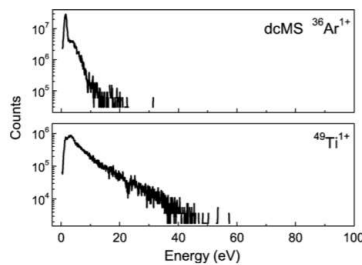


Fig. 2. The ion energy distributions taken from a conventional DC magnetron discharge. The Ar pressure was 0.4 Pa, the applied power 1 kW, and the target material was Ti. The recorded counts have been adjusted with the corresponding isotope abundance.

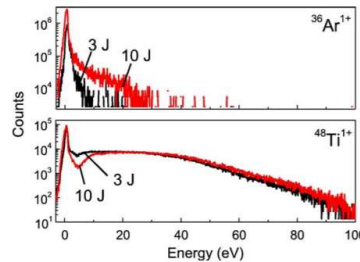
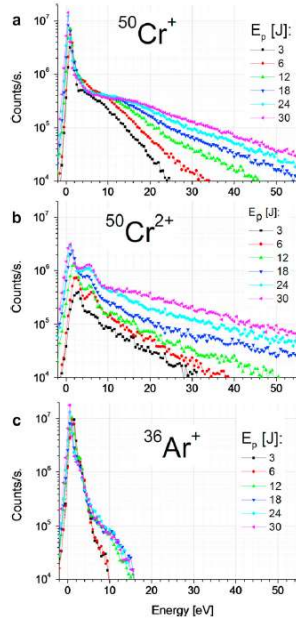


Fig. 3. The ion energy distributions for Ar<sup>1+</sup> and Ti<sup>1+</sup> ions measured from a HiPIMS discharge. The Ar pressure was 0.4 Pa, the pulse energy 3 and 10 J, and the target was made of Ti. The recorded counts have been adjusted with the corresponding isotope abundance.

128

### Higher-energy metal ions



### Ion Energy Distribution Functions

high-energy tails due to

- gas rarefaction
- potential humps of associated with traveling ionization zones – next slide

G. Greczynski, L. Hultman, Vacuum 84 (2010) 1159

129

129

### Higher-energy metal ions

APPLIED PHYSICS LETTERS 103, 144103 (2013)

### Drifting potential humps in ionization zones: The “propeller blades” of high power impulse magnetron sputtering

André Anders,<sup>1,a)</sup> Matjaž Panjan,<sup>1,2</sup> Robert Franz,<sup>1,3</sup> Joakim Andersson,<sup>1,4</sup> and Pavel Ni<sup>1</sup>

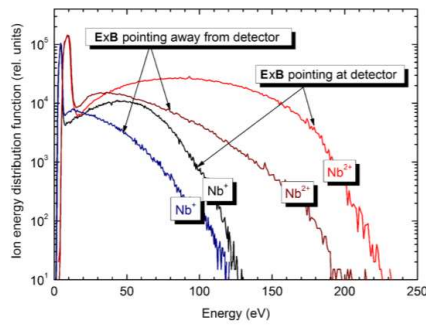
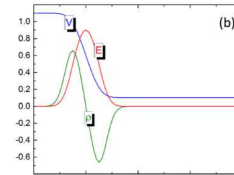
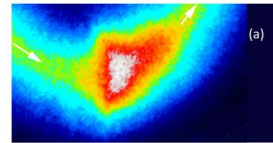


FIG. 2. Ion energy distribution function measured with a Hidden EQP300 energy analyzer at 140 mm distance from the target, recording niobium ions emitted near the target plane tangentially from the racetrack; discharge voltage 350 V, 200 μs pulse length, 100 pulses per second, 275 A peak, in 0.53 Pa of Ar.



$$E_i = E_0 + \frac{m_i}{2} v_{iZ}^2 + eQV_{hump}, \quad (8)$$

where  $E_0$  represents the energy from the sputtering and collisional processes,  $v_{iZ}$  is the drift velocity of the ionization zone,  $Q$  is the ion charge state number, and  $V_{hump}$  is the height of the potential hump. Ions moving in the opposite



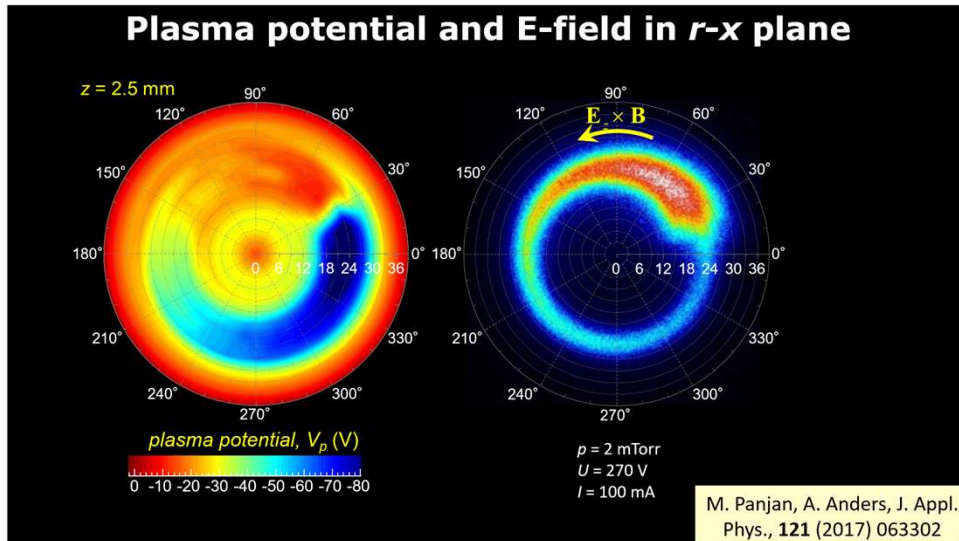
André Anders

Potential humps – double layer

130

130





### Further reading on spokes and high energy metal ions in HiPIMS

IOP Publishing  
J. Phys. D: Appl. Phys. **47** (2014) 224002 (13pp)  
doi:10.1088/0022-3727/47/22/224002

#### Origin of the energetic ions at the substrate generated during high power pulsed magnetron sputtering of titanium

**C Maszl, W Breilmann, J Benedikt and A von Keudell**  
Research Department Plasmas with Complex Interactions, Ruhr-Universität Bochum, Institute for Experimental Physics II, D-44780 Bochum, Germany

Plasma Chemistry and Plasma Processing (2020) 40:643–660  
<https://doi.org/10.1007/s11090-019-10052-3>

ORIGINAL PAPER

#### Pattern Formation in High Power Impulse Magnetron Sputtering (HiPIMS) Plasmas

Julian Held<sup>1</sup> · Achim von Keudell<sup>1</sup>

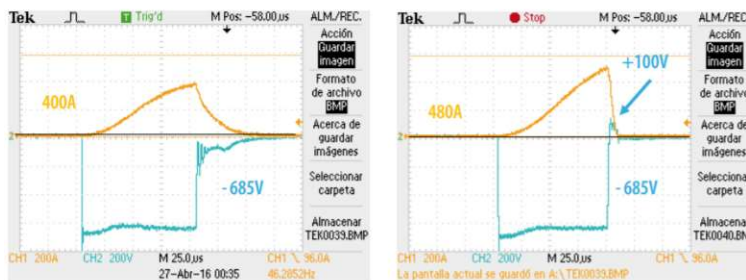
The IEDF of metal species arriving at the substrate defines the energy input during film growth, which is of paramount importance for all film properties. Since the DL around the spokes provides a mechanism to overcome the return effect in HiPIMS and to provide energetic  $Ti^+$  species, one may conclude that the spoke phenomenon is not a nuisance or peculiarity of the HiPIMS process, but rather the *essence* of HiPIMS plasmas explaining their good performance for material synthesis applications.

## Bipolar pulsed high-power impulse magnetron sputtering

133

## Bipolar pulsed high-power impulse magnetron sputtering

G.Eichenhofer, I Fernandez, A Wennberg  
Vakuum in Forschung und Praxis · April 2017



Ivan Fernandez: Nano4Energy

- Novel HIPIMS power supply the hiPV, hiPlus option
- positive voltage reversal for **positive ion assisted deposition on insulating substrates with reducing the tendency to arcing.**

**nano4ENERGY**

134

134

## Bipolar pulsed high-power impulse magnetron sputtering

### Increased deposition rate

Wu, Haehnlein, Shchelkanov, McLain, Patel, Uhlig, Jurczyk, Leng, Ruzic, *Vacuum*, 150, 216 (2018)

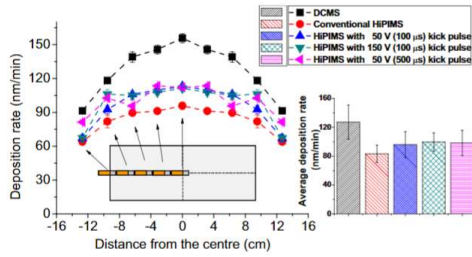
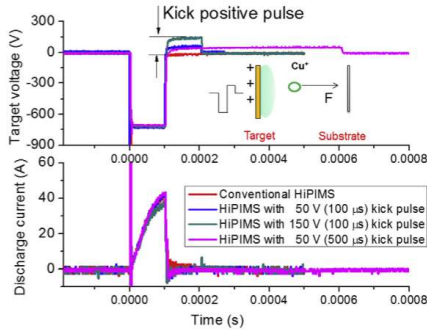


Fig. 3. Deposition rate of Cu films across the center-line length of Cu target. Average deposition rate is the average value of deposition rate of all the samples at different position.



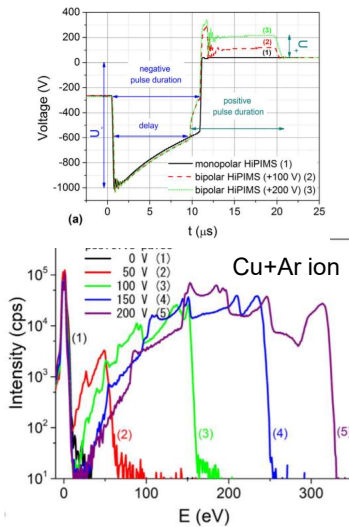
135

135

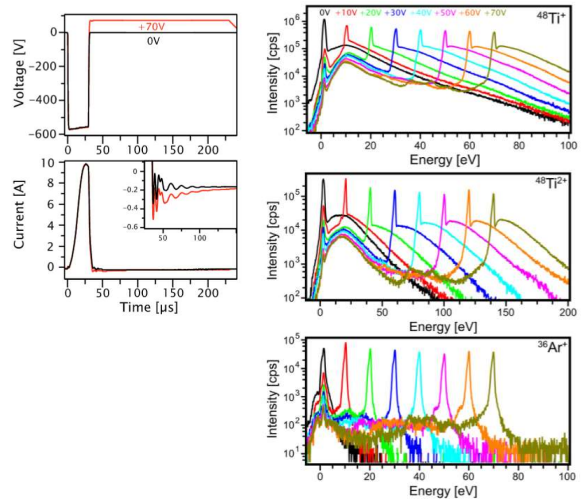
## Bipolar pulsed high-power impulse magnetron sputtering

### Precise control of ion energy

Velicua, Ianoș, Porosnicu, Mihăilă, Burducea, Velea, Cristea, Munteanu, Tiron, *SCT* 359, (2019), 97-107



Keraudy, Viloin, Raadu, Brenning, Lundin, Helmersson, *SCT* 359, (2019), 433



136



# Experimental verification of deposition rate increase, with maintained high ionized flux fraction, by shortening the HiPIMS pulse

Plasma Sources Sci. Technol. **30** (2021) 045006 (8pp)

T Shimizu<sup>1,2</sup>, M Zanaška<sup>1</sup>, R P Villoian<sup>1</sup>, N Brenning<sup>1,3</sup>, U Helmersson<sup>1</sup> and Daniel Lundin<sup>1,4,\*</sup>

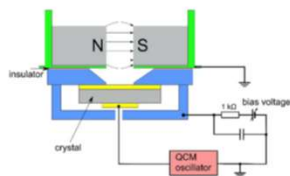


Figure 5. Schematic of the gridless ion meter. Impinging electrons are repelled by a local magnetic field, which prevents the electrons from reaching the biased top QCM electrode.

Plasma Sources Sci. Technol. **30** (2021) 045006

T Shimizu et al

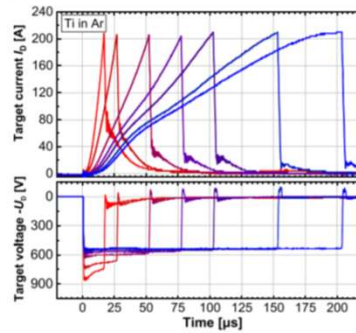


Figure 1. The applied discharge voltage  $U_D$  (lower panel) and the resulting discharge current ( $I_D$ ) (upper panel) waveforms of the Ti/Ar discharge when varying the pulse length in the range 15–200  $\mu\text{s}$ . The discharge voltage was adjusted to maintain a peak current of 200 A ( $U_{D,peak} = 1.10 \text{ A cm}^{-2}$ ). The pulse repetition frequency was adjusted to maintain a constant time-averaged power of 1 kW.

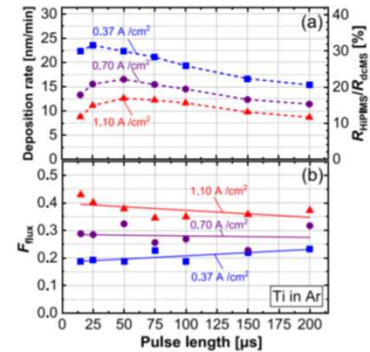


Figure 2. (a) Deposition rate (left axis) with corresponding normalized HiPIMS deposition rate to the dcMS deposition rate,  $R_{HiPIMS}/R_{dcMS}$  (right axis), and (b) Ionized flux fraction  $F_{flux}$  as a function of pulse length for three different  $J_{D,peak}$  of 0.37, 0.70 and 1.10  $\text{A cm}^{-2}$ . An averaged power of 1 kW is maintained constant independent of pulse length and  $J_{D,peak}$ .

137

## Tutorial: Reactive high power impulse magnetron sputtering (R-HiPIMS)

Cite as: J. Appl. Phys. **121**, 171101 (2017); <https://doi.org/10.1063/1.4978350>

Submitted: 17 November 2016 . Accepted: 18 February 2017 . Published Online: 21 March 2017

André Anders

JOURNAL OF APPLIED PHYSICS **121**, 080901 (2017)



### Perspective: Is there a hysteresis during reactive High Power Impulse Magnetron Sputtering (R-HiPIMS)?

K. Strijckmans, F. Moens, and D. Depla

Department of Solid State Sciences, Ghent University, Krijgslaan 281(S1), 9000 Gent, Belgium

138

138

## Smooth composition control of oxynitrides - TaON (HIPIMS with feedback pulsed reactive gas flow control (RGFC))

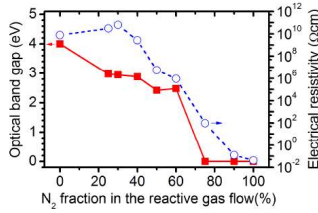
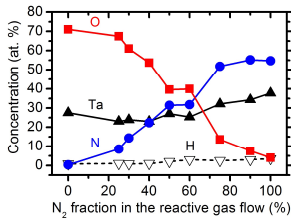
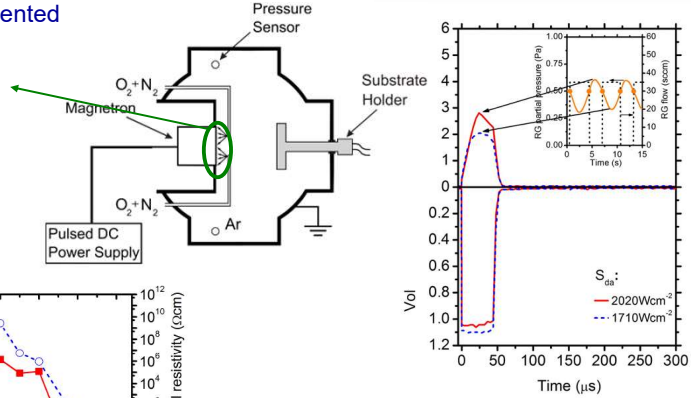
J. Rezek, J. Vlcek, J. Houska, R. Cerstvy, *Thin Solid Films* 566, 70 (2014)

Reactive gas inlets in front of the target and oriented towards the target + dense HIPIMS plasma

↓  
dissociation of  $O_2$  and  $N_2$

~~different reactivities of  $O_2$  and  $N_2$~~

comparable reactivities of O and N



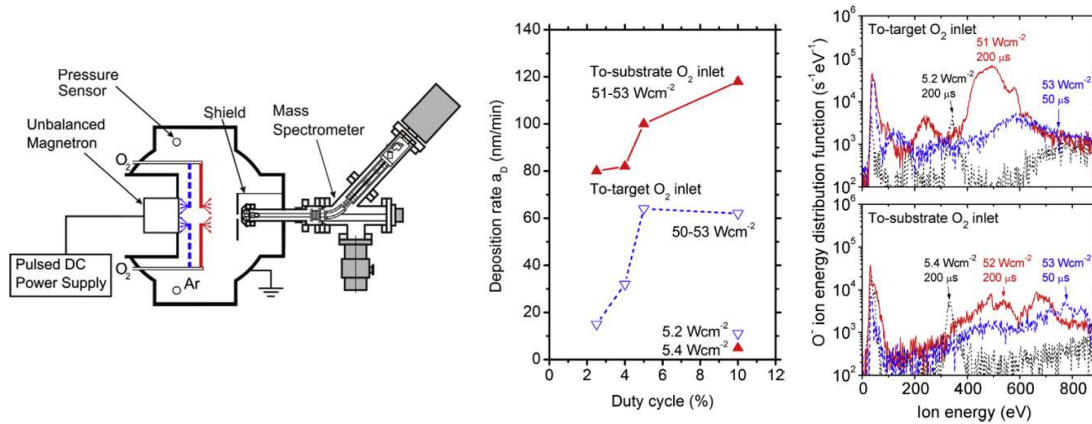
139

## Benefits of the controlled reactive high-power impulse magnetron sputtering of stoichiometric $ZrO_2$ films

J. Vlcek\*, J. Rezek, J. Houška, T. Kozák, J. Kohout *Vacuum* 114 (2015) 131–141

*Department of Physics and NTIS, European Centre of Excellence, University of West Bohemia, Univerzitní 8, 30614 Plzeň, Czech Republic*

Major advance: HIPIMS with feedback pulsed reactive gas flow control (RGFC)



140

## Bibliography

(vacuum, plasmas)

- P.A. Redhead, J.P. Hobson, and E.V. Kornelsen, *The Physical Basis of Ultrahigh Vacuum*, Chapman and Hall, London, 1968.
- A. Neugebauer, "Condensation, Nucleation, and Growth of Thin Films," in *Handbook of Thin Film Technology*, edited by L.I. Maissel and R. Glang, McGraw-Hill, New York 1970.
- J.E. Greene, "Physics of Film Growth from the Vapor Phase," in *Multicomponent and Multilayered Thin Films for Advanced Technologies: Techniques, Fundamentals, and Devices*, ed. by O. Auciello, NATO Advanced Study Institute, Kluwer Academic Publishers, Boston 1993, p. 39.
- R. Glang, "Vacuum Evaporation," in *Handbook of Thin Film Technology*, edited by L. I. Maissel and R. Glang, McGraw-Hill, NY (1970)..
- G.K. Wehner and G.S. Anderson, "The Nature of Physical Sputtering" in *Handbook of Thin Film Technology*, edited by L.I. Maissel and R. Glang, McGraw-Hill, NY 1970.
- B. Chapman, *Glow Discharge Processes*, Wiley Interscience, New York 1980.
- J.A. Thornton and J.E. Greene, "Plasmas in Deposition Process," in *Deposition Technologies for Films and Coatings*, edited by R.F. Bunshah, Noayes Publications, Park Ridge, New Jersey 1994, p. 29.
- S.M. Rossnagel, "Magnetron Plasma Deposition Processes," in *Handbook of Plasma Processing Technology*, edited by S.M. Rossnagel, J.J. Cuomo, and W.D. Westwood Noyes Publications, Park Ridge, New Jersey 1990.
- K.H. Müller, "Film Growth Modification by Concurrent Ion Bombardment: Theory and Simulation," in *Handbook of Ion-Beam Processing Technology*, edited by J.J. Coumo, S.M. Rossnagel, and H.R. Kaufman, Noyes Publications, Park Ridge, New Jersey 1989, p. 241.

141

141

## Bibliography

(sputtering, sputter deposition)

- P. Sigmund, "Theory of Sputtering I. Sputtering Yield of Amorphous and Polycrystalline Targets," *Physical Review* 184, 383 (1969).
- P.Sigmund, "Sputtering by ion bombardment: theoretical concepts", in *Sputtering by particle bombardment I*, edited by R. Behrish, Springer-Verlag, 1981
- HH Anderson and HL Bay, "Sputtering yield measurements", in *Sputtering by particle bombardment I*, edited by R. Behrish, Springer-Verlag, 1981
- G.K. Wehner and G.S. Anderson, "The Nature of Physical Sputtering" in *Handbook of Thin Film Technology*, edited by L.I. Maissel and R. Glang, McGraw-Hill, NY 1970.
- J.L. Vossen and J.J. Cuomo, "Glow Discharge Sputter Deposition," in *Thin Film Processes*, edited by J.L. Vossen and W. Kern, Academic Press, NY 1978.
- J.A. Thornton and A.S. Penfold, "Cylindrical Magnetron Sputtering," in *Thin Film Processes*, edited by J.L. Vossen and W. Kern, Academic Press, NY 1978.
- R.K. Waits, "Planar Magnetron Sputtering," in *Thin Film Processes*, edited by J.L. Vossen and W. Kern, Academic Press, New York 1978.
- B. Chapman, *Glow Discharge Processes*, Wiley Interscience, New York 1980.
- J.A. Thornton and J.E. Greene, "Plasmas in Deposition Processes," in *Deposition Technologies for Films and Coatings*, edited by R.F. Bunshah, Noayes Publications, Park Ridge, New Jersey 1994, p. 29.
- S.M. Rossnagel, "Magnetron Plasma Deposition Processes," in *Handbook of Plasma Processing Technology*, edited by S.M. Rossnagel, J.J. Cuomo, and W.D. Westwood Noyes Publications, Park Ridge, New Jersey 1990.

142

142

## Part 1. Fundamentals of Sputter Deposition

### 1.1 Elements of kinetic theory of gases

- Gas laws
- mean free path
- gas impingement rate

### 1.2 Elements of plasma physics

- Plasma probes
- Sheath width
- Penning ionization
- Electron energy distribution functions

### 1.3 Glow discharge maintenance

- Secondary ion-electron emission
- Electron ionization cross-sections

### 1.4 Sputtering yield

- Linear cascade model
- Correction for threshold effects
- Sputtering efficiency
- Energy of sputtered atoms
- Other energetic particles: backscattered ions and negative oxygen ions

### 1.5 Transport in the gas phase

- Thermalization
- Deposition rate calculation

### 1.6 Sputtering systems

- Magnetron sputtering
- Reactive sputtering

### 1.7 HIPIMS

- Source of metal ions
- Time separation between gas and metal ions
- High energy ions
- Lower deposition rates
- Bipolar HIPIMS
- Control of doubly charged ions